

Strain-modulated bulk photovoltaic effect in negative piezoelectrics

Yue Gao¹, Wenli Zou¹, Jian Zhou^{2,*}, Chunmei Zhang^{1,3,#}¹School of Physics, Northwest University, Xi'an 710127, China²Center for Alloy Innovation and Design, State Key Laboratory for Mechanical Behavior of Materials, Xi'an Jiaotong University, Xi'an 710049, China³Shaanxi Key Laboratory for Theoretical Physics Frontiers, Xi'an 710127, China

Corresponding authors. E-mail: *jianzhou@xjtu.edu.cn, #chunmeizhang@nwu.edu.cn

Received December 19, 2025; accepted March 27, 2026

Supplemental Material

I. Tight-binding model of 1D chain

The simple 1D model has two ions in a unit cell. The formal charges of them are $\pm Z$, and at the equilibrium structure, their interatomic bond distances are denoted as R_1 and R_2 , giving lattice constant of $L = R_1 + R_2$. To simulate their ionic interactions, we adopt spring-like potential, described by spring stiffness coefficients k_1 and k_2 , respectively. Equal separation structure is considered as the referenced structure with zero polarization. Under a nonzero distance mismatch $\eta = R_2 - R_1$, net polarization occurs, which is contributed from ionic and electronic parts separately. In the following, we breakdown their contributions and estimate their effects on the sign of piezoelectric coefficient $d = \frac{\delta P}{\delta \varepsilon}$. Here, ε represents a uniaxial strain along the chain, defined as $\varepsilon = \frac{L' - L}{L}$, where L' and L are the lattice constants in the strained and equilibrium states. The ionic part contribution can be written as

$$P_{\text{ion}} = \frac{1}{L} \left(-Z \times -\frac{\eta}{2} \right) = \frac{Z\eta}{2L} \quad (\text{S1})$$

Under a strain, the new lattice constant $L' = R'_1 + R'_2$. At such a new equilibrium structure, it is clearly that $\frac{R'_1 - R_1}{R'_2 - R_2} = \frac{k_2}{k_1}$. Hence, we can estimate the piezoelectric constant as

$$\begin{aligned} d_{\text{ion}} &= \frac{\delta P_{\text{ion}}}{\delta \varepsilon} = Z \frac{\frac{\eta'}{2L'} - \frac{\eta}{2L}}{L' - L} L = \frac{ZL}{2(L' - L)} \frac{(L' - L) \left(-\eta + \frac{k_1 - k_2}{k_1 + k_2} L \right)}{LL'} \\ &= \frac{Z}{2L'} \frac{[-\eta(k_1 + k_2) + L(k_1 - k_2)]}{k_1 + k_2} = \frac{Z}{2L'} \frac{[-\eta(k_1 + k_2) + L(k_1 - k_2)]}{k_1 + k_2} \\ &= \frac{Z}{(k_1 + k_2)L'} (R_1 k_1 - R_2 k_2) \end{aligned} \quad (\text{S2})$$

Therefore, the sign of $\frac{\delta P_{\text{ion}}}{\delta \varepsilon}$ is determined by two factors, the formal charge Z and the relative magnitude between equilibrium distance $\frac{R_2}{R_1}$ and stiffness $\frac{k_1}{k_2}$.

Next, we discuss the electronic polarization P_{el} . As described in the main text, we adopt a two-orbital spinless tight-binding model

$$H = -\frac{t}{2} \sum_n (c_n^\dagger c_{n+1} + c_{n-1}^\dagger c_n) + \frac{\Delta}{2} \sum_n (-1)^i c_n^\dagger c_n \quad (\text{S3})$$

Here, t is the hopping integral between nearest neighbor ions, and $\pm \frac{\Delta}{2}$ is their on-site energies. Giving $\beta = \left| \frac{t}{\Delta} \right| \ll 1$, the eigenenergies take the form of

$$E_{\pm}(k) = \pm \frac{\Delta}{2} \sqrt{1 + 16\beta^2 \cos^2 \frac{kL}{2}} \simeq \pm \frac{\Delta}{2} \left(1 + 8\beta^2 \cos^2 \frac{kL}{2}\right) \quad (S4)$$

It opens a direct band gap ($E_g = \Delta$) at the X point. Through $(H(k) - E(k)I)\psi(k) = 0$, the conduction band Bloch wave function are derived as:

$$|u_+(k)\rangle = \frac{1}{\sqrt{N_+(k)}} \begin{pmatrix} -\beta g(k)\Delta \\ \frac{\Delta}{2} \left(1 - \sqrt{1 + 4\beta^2 |g(k)|^2}\right) \end{pmatrix} \quad (S5)$$

And the valence band Bloch wave function is:

$$|u_-(k)\rangle = \frac{1}{\sqrt{N_-(k)}} \begin{pmatrix} -\beta g(k)\Delta \\ \frac{\Delta}{2} \left(1 + \sqrt{1 + 4\beta^2 |g(k)|^2}\right) \end{pmatrix} \quad (S6)$$

where $g(k) = e^{-iR_1 k} + e^{iR_2 k} = 2e^{i\frac{\eta}{2}k} \cos \frac{Lk}{2}$, and $\sqrt{N_{\pm}(k)}$ is the normalization factor.

Perform a Taylor expansion about β : $|u_-(k)\rangle \simeq \begin{pmatrix} -\beta g(k) + O(\beta^3) \\ 1 + O(\beta^2) \end{pmatrix} \simeq \begin{pmatrix} -\beta g(k) \\ 1 \end{pmatrix}$.

This gives

$$i \left\langle u_-(k) \left| \frac{d}{dk} \right| u_-(k) \right\rangle = i\beta^2 g^*(k) \frac{dg(k)}{dk} \quad (S7)$$

Here,

$$g^*(k) = 2e^{-i\frac{\eta}{2}k} \cos \frac{Lk}{2} \quad (S8)$$

$$\frac{dg(k)}{dk} = e^{i\frac{\eta}{2}k_x} \left(i\eta \cos\left(\frac{Lk_x}{2}\right) - L \sin\left(\frac{Lk_x}{2}\right) \right) \quad (S9)$$

$$|g(k)|^2 = 4\cos^2 \frac{Lk}{2} \quad (S10)$$

$$\frac{d}{dk} |g(k)|^2 = -4L \cos\left(\frac{Lk_x}{2}\right) \sin\left(\frac{Lk_x}{2}\right) \quad (S11)$$

Then, we can obtain

$$ig^*(k)g'(k) = -2\eta \cos^2 \frac{Lk}{2} - 2iL \cos\left(\frac{Lk_x}{2}\right) \sin\left(\frac{Lk_x}{2}\right) \quad (S12)$$

$$\begin{aligned} P_{\text{el}} &= \frac{ie}{2\pi} \int_{-\frac{\pi}{L}}^{\frac{\pi}{L}} \left\langle u_-(k) \left| \frac{d}{dk} \right| u_-(k) \right\rangle dk = \frac{ie}{2\pi} \int_{-\frac{\pi}{L}}^{\frac{\pi}{L}} \beta^2 g^*(k) g'(k) dk \\ &= -\frac{e\beta^2 \eta}{4\pi} \int_{-\frac{\pi}{L}}^{\frac{\pi}{L}} |g(k)|^2 dk + \frac{e\delta^2 i}{2\pi} \int_{-\frac{\pi}{L}}^{\frac{\pi}{L}} \frac{d}{dk} |g(k)|^2 dk \\ &= -\beta^2 \frac{e\eta}{L} + O(\beta^3) \end{aligned} \quad (S13)$$

Here, we expand the expression according to the power of unit-less parameter β . One sees that the polarization is zero when the two sites are equally separated ($\eta = 0$), consistent with previous assumption. Since $P_{\text{el}} \propto \frac{1}{L}$, the tensile (compressive) strain could reduce (enhance) the magnitude of $|P_{\text{el}}|$. Performing derivative, we have

$$\begin{aligned}
d_{el} &= \frac{\delta P_{el}}{\delta \varepsilon} = -\beta^2 e \frac{\frac{\eta'}{L'} - \frac{\eta}{L}}{L' - L} L = \frac{-\beta^2 e L (L' - L) \left(-\eta + \frac{k_1 - k_2}{k_1 + k_2} L \right)}{L L'} = \frac{-\beta^2 e [-\eta(k_1 + k_2) + L(k_1 - k_2)]}{L' (k_1 + k_2)} \\
&= \frac{-2\beta^2 e}{(k_1 + k_2)L'} (R_1 k_1 - R_2 k_2) \quad (S14)
\end{aligned}$$

Therefore, the sign of $\frac{\delta P_{el}}{\delta \varepsilon}$ is also determined by the relative magnitude between equilibrium distance $\frac{R_2}{R_1}$ and stiffness $\frac{k_1}{k_2}$. Note that we assume a constant hopping integral t in the model, which is only approximately correct under small strain. Nevertheless, as $P_{el} \propto \beta^2$, the electron contributed polarization (and the piezoelectric coefficient) is order of magnitude smaller than the ion contributions P_{ion} .

We assume $Z > 0$, under the condition of $\frac{k_1}{k_2} < \frac{R_2}{R_1}$, this model exhibits a negative piezoelectric response. Then, we assume specific TB parameters to realize the negative piezoelectricity. Building upon reported negative piezoelectric materials [S1, S2] with negative longitudinal piezoelectric coefficients, we choose suitable atomic distances ($R_1 = 1.5 \text{ \AA}$ and $R_2 = 2.2 \text{ \AA}$). The relationship between "stiff" k_1 and "soft" k_2 should satisfy $k_1 \frac{R_1}{R_2} < k_2$. In addition, to maintain a clear distinction between the "stiff" and "soft" regions, we consider $k_2 < k_1$. These two criteria are sufficient conditions for the negative piezoelectricity in the 1D chain model. Here, we take a representative value of $k_1 = 1.2k_2$ to perform our analysis. Note that this specific choice would not affect the main physical conclusion of this work. t_0 is the hopping constant between the nearest neighbor, which is set to be 0.2 eV. Collectively, these parameters conform to the requirement $\frac{\delta P}{\delta \varepsilon} < 0$ and successfully demonstrate negative piezoelectricity, as shown in Table S1. Note that to capture the strain-dependent evolution of nonlinear optical response, a decay length r_0 of 1.433 \AA was incorporated for the hopping parameter when shift current is calculated. Here, the hopping parameter are evaluated by $t_i(R_i) = t_0 \exp\left(-\frac{R_i - R_0}{r_0}\right)$.

TABLE S1. The strain ($\varepsilon = (L' - L)/L$), the lattice parameters (L'), the distance mismatch (η), the polarization magnitude P_{ion} , P_e , and P_{tot} , band gap, and the largest peak value of σ^{xxx} of 1D chain under strain in the negative piezoelectric regime.

ε	$L'(\text{\AA})$	$\eta(\text{\AA})$	P_{ion} (10^{-2} e/\AA)	P_e (10^{-3} e/\AA)	P_{tot} (e/\AA)	Band gap (eV)	σ^{xxx} ($\mu\text{A/V}^2$)
-2%	3.626	0.694	9.55	7.65	0.1033	1.0123	16.675
-1 %	3.663	0.697	9.50	7.60	0.1027	1.0121	16.564
0%	3.700	0.700	9.45	7.57	0.1021	1.0119	16.438
1%	3.737	0.703	9.41	7.53	0.1016	1.0117	16.296
2%	3.774	0.706	9.36	7.49	0.1011	1.0115	16.139

Furthermore, based on the above derivation, the condition for realizing a positive piezoelectric response can be correspondingly obtained as $k_1 > \frac{R_2}{R_1} k_2$. Within the same parameter framework, stiffness coefficients satisfying this condition are adopted to model the shift current response in positive piezoelectric materials. As a representative case, we choose $k_2 = 0.5 k_1$ to investigate the behavior. One sees that tensile strain concurrently increases polarization and shift vector, yielding an enhanced shift current component.

TABLE S2. The strain ($\varepsilon = (L' - L)/L$), the lattice parameters (L'), the distance mismatch (η), the polarization magnitude P_{ion} , P_e , and P_{tot} , band gap, and the largest peak value of σ^{xxx} of 1D chain under strain in the positive piezoelectric regime.

ε	$L'(\text{\AA})$	$\eta(\text{\AA})$	P_{ion} ($10^{-2} e/\text{\AA}$)	P_e ($10^{-3} e/\text{\AA}$)	P_{tot} ($e/\text{\AA}$)	Band gap (eV)	σ^{xxx} ($\mu\text{A}/\text{V}^2$)
-2%	3.626	0.694	9.27	7.41	0.1000	1.0115	16.210
-1 %	3.663	0.697	9.36	7.49	0.1011	1.0117	16.334
0%	3.700	0.700	9.46	7.57	0.1021	1.0119	16.441
1%	3.737	0.703	9.55	7.64	0.1032	1.0121	16.533
2%	3.774	0.706	9.65	7.77	0.1042	1.0122	16.608

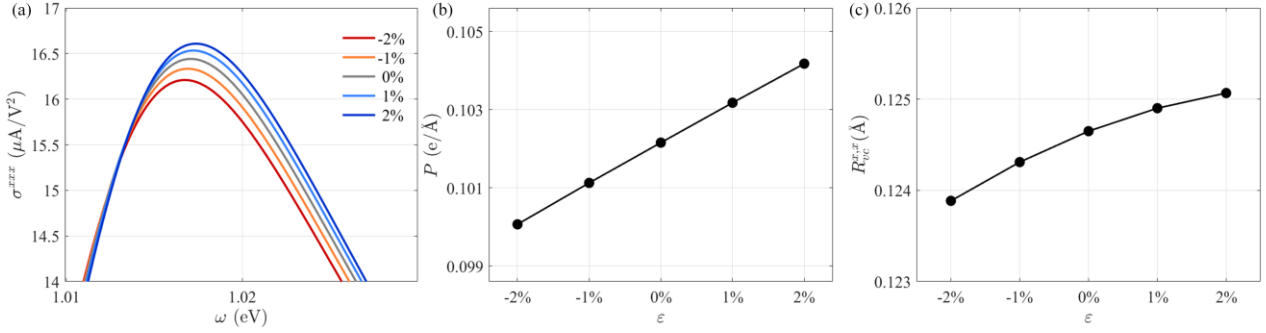


Figure S1 (a) Shift current photoconductance, (b) polarization and (c) shift vector at X points of the 1D chain TB model under strain ($\varepsilon = -2\% \sim 2\%$) in the positive piezoelectric regime.

TABLE S3. Comparison of electric polarization P , BPVE response evolution for positive and negative piezoelectrics under compressive strain. The symbols \uparrow (\downarrow) denote increase (decrease) trends under strain, respectively.

	materials	Compressive strain	
		P	BPVE response
negative piezoelectrics	r-GeX	\uparrow	\uparrow
	group-V monolayers[S3]	\uparrow	\uparrow
positive piezoelectrics	2H-MoS ₂ [S4]	\downarrow	\downarrow
	3R-MoS ₂ [S5]	\downarrow	\downarrow

II. Electronic structure of r-GeTe

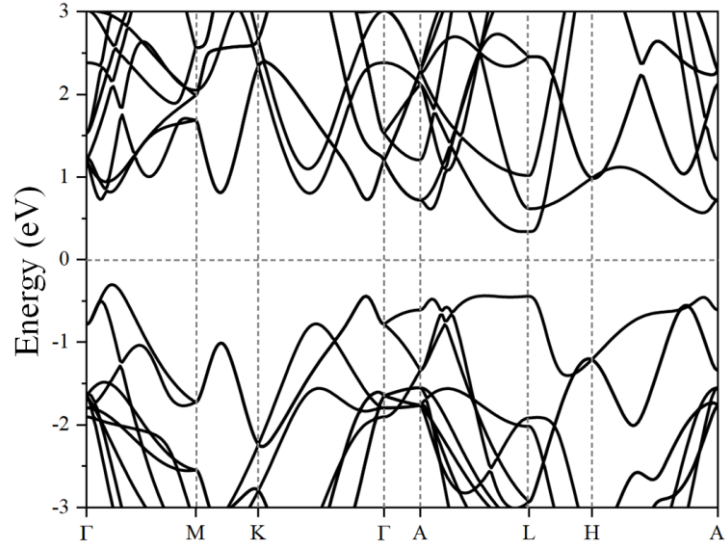


Figure S2. Calculated band structure without spin-orbit coupling (SOC) of r-GeTe.

III. The resolved shift current of r-GeTe

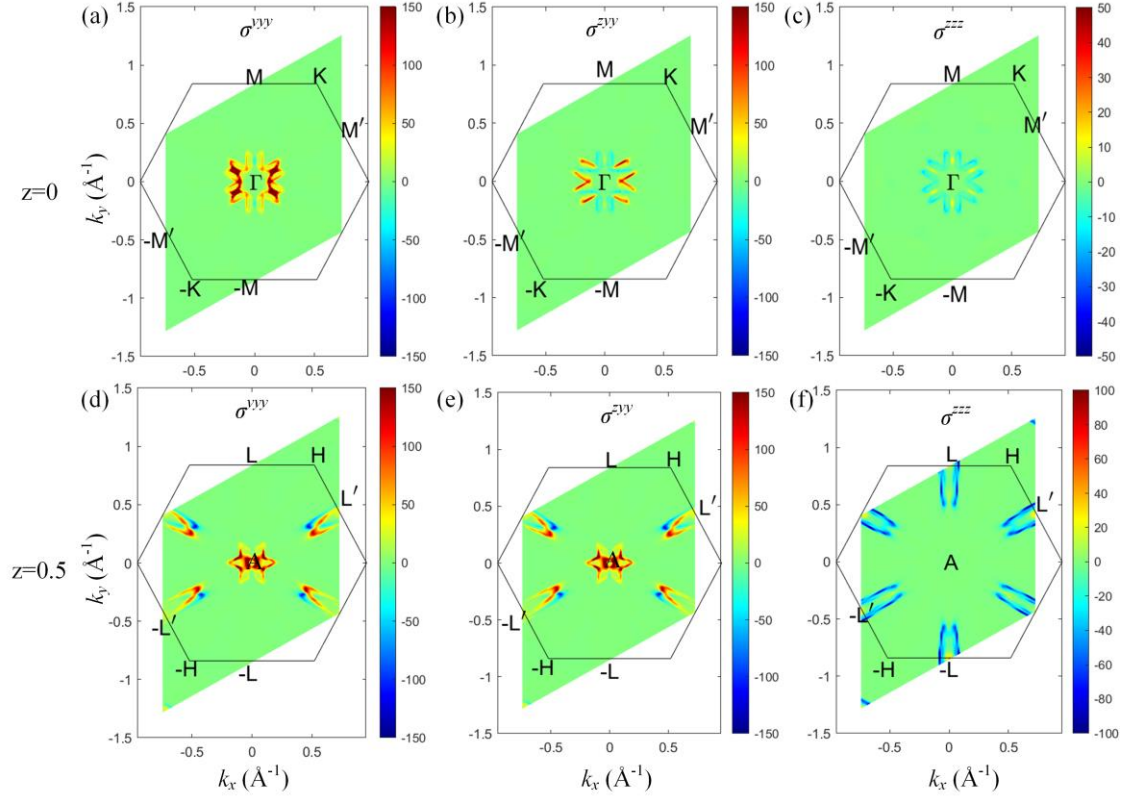


Figure S3. The k -resolved shift current conductance σ^{yyy} , σ^{zyy} , and σ^{zzz} in the first Brillouin zone (BZ) of the bulk r-GeTe at $k_z = 0 \times \pi/c$, $k_z = 0.5 \times \pi/c$.

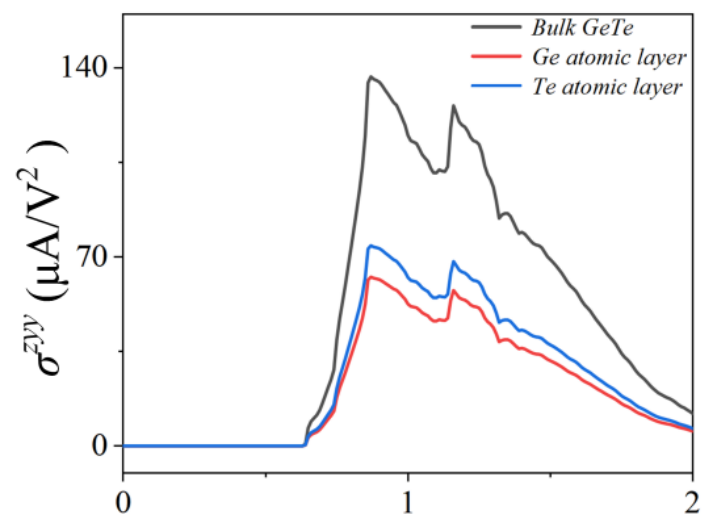


Figure S4. The calculated σ^{xy} photoconductivity on the Ge atomic layer and Te atomic layer.

IV. k -resolved shift vector and r_{mn} of r-GeTe under strain

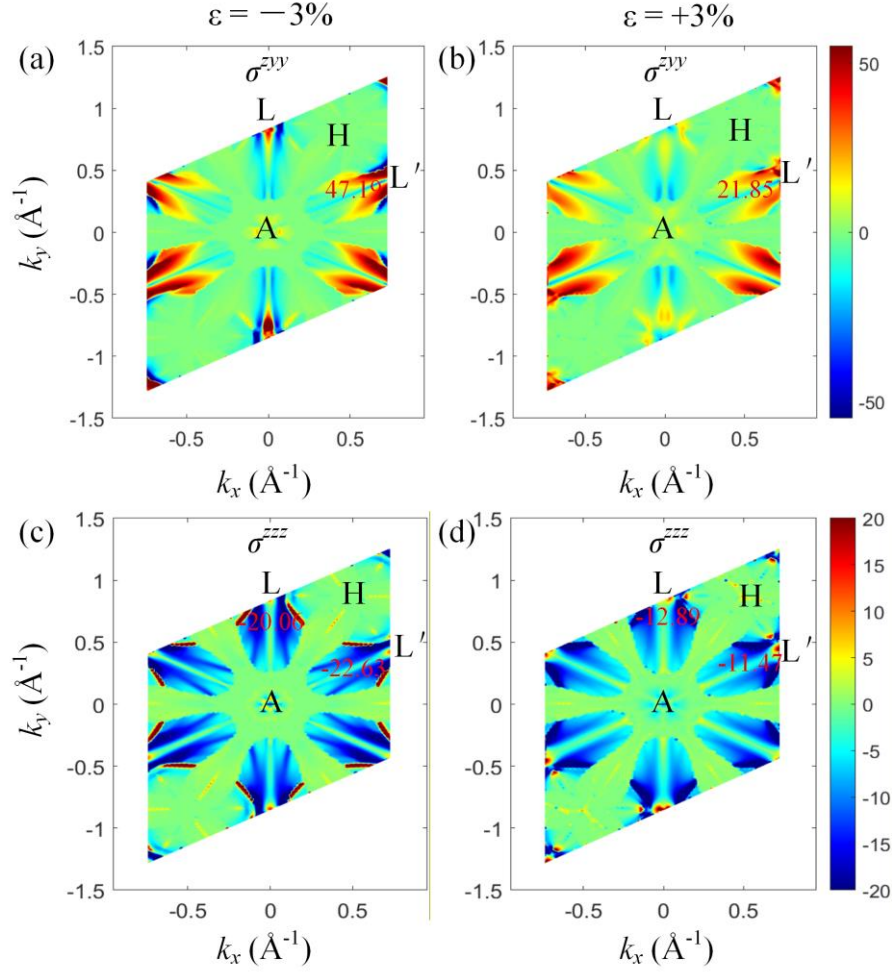


Figure S5. k -resolved shift vector of σ^{zyy} and σ^{zzz} in the first BZ between the valence band maximum (VBM) and the conduction band minimum (CBM), CB-2 of r-GeTe under out-of-plane strain ($\varepsilon = -3\%$, 3%) at $k_z = 0.5 \times \pi / c$. The typical values around L and L' points (0.5, 0, 0.5) for all cases are marked in each panel, which are (a) 47.19 Å, (b) 21.85 Å, (c) -20.06 Å, -22.63 Å, (d) -12.89 Å, -11.47 Å, respectively. One sees that the shift vector exhibits a larger magnitude under compressive strain compared to tensile strain.

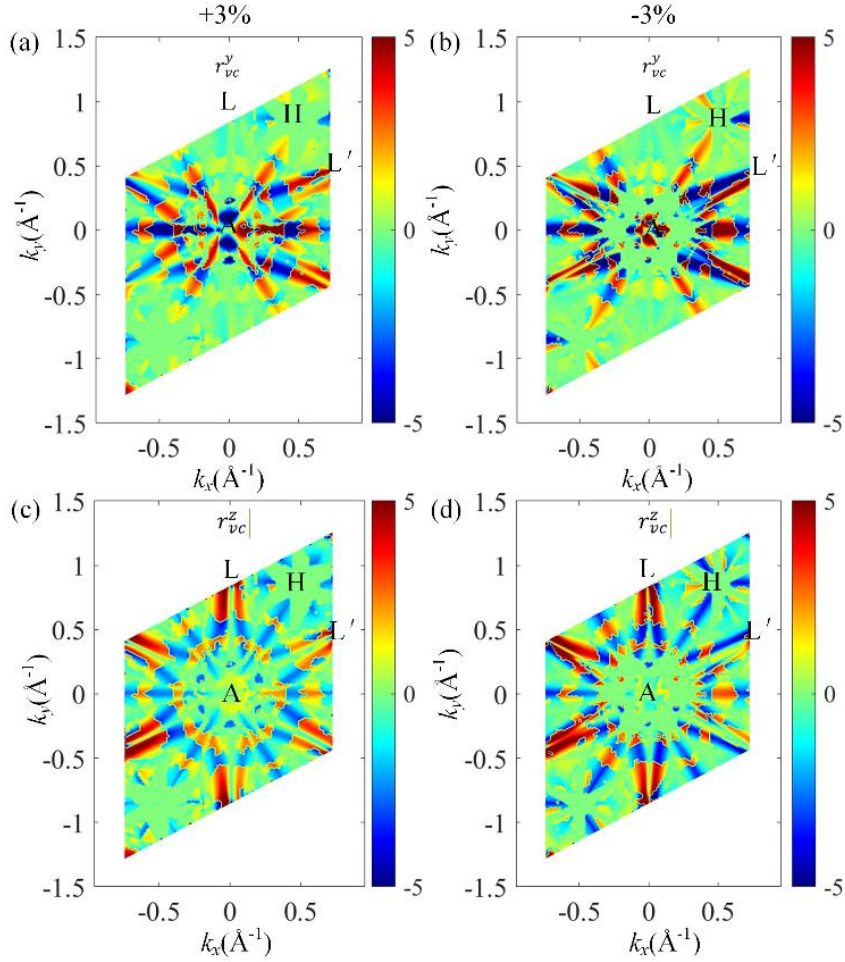


Figure S6 The interband Berry connection r_{vc}^y and r_{vc}^z of r-GeTe in the BZ between the top valence band and bottom conduction band under $\pm 3\%$ strain, at $k_z = 0.5 \times \pi/c$. The typical values of r_{vc}^y around the L' point (0.5, 0, 0.5) for the cases are (a) 2.15 Å, (b) 5.23 Å, and that of r_{vc}^z around the L point (0, 0.5, 0.5) for the cases are (c) 1.76 Å, (d) 4.66 Å, respectively. One sees that the interband Berry connection r_{vc}^y and r_{vc}^z exhibit a larger magnitude under compressive strain compared to the tensile strain.

TABLE S4 The s -, p -orbital weight of Ge and Te atoms in VBM and CBM.

	Band	Ge- s	Ge- p_y	Ge- p_z	Ge- p_x	Te- s	Te- p_y	Te- p_z	Te- p_x
-3%	VBM	0.112	0.012	0.107	0.012	0.024	0.080	0.114	0.025
	CBM	0.018	0.104	0.073	0.104	0.031	0.043	0.059	0.044
+3%	VBM	0.169	0.005	0.071	0.005	0.073	0.008	0.203	0.008
	CBM	0.002	0.170	0.020	0.170	0.005	0.036	0.008	0.036

V. The shift current of r-GeTe under strain

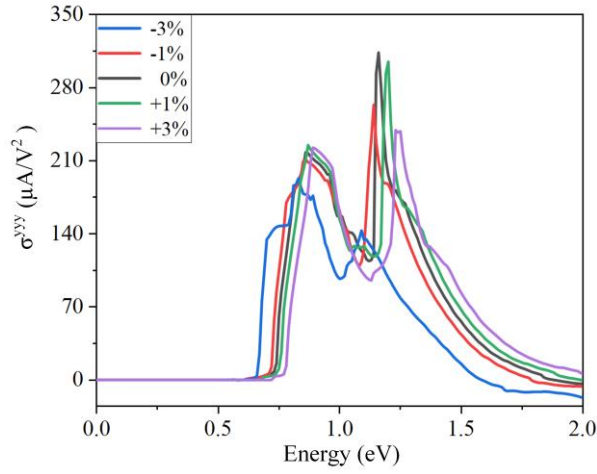


Figure S7. Shift current conductance σ^{yyy} of the bulk r-GeTe under out-of-plane strain.

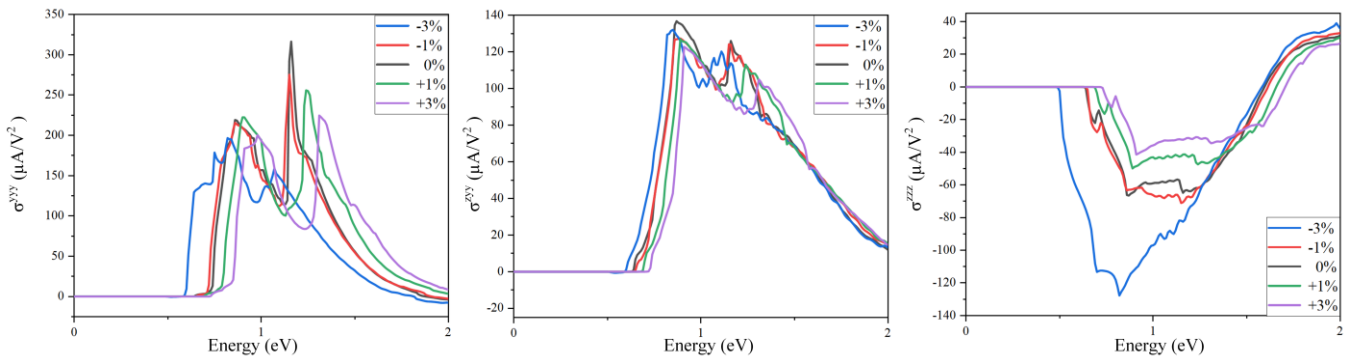


Figure S8. Shift current conductance (a) σ^{yyy} , (b) σ^{zyy} , and (c) σ^{zzz} of the bulk r-GeTe under in-plane biaxial strain.

VI. The shift current of r-GeX (X=S, Se)

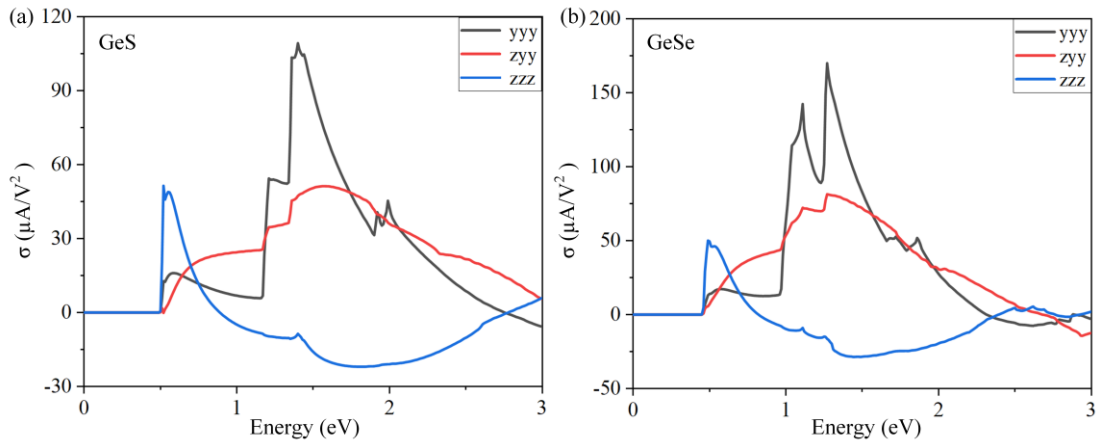


Figure S9. Shift current conductance of the bulk rhombohedral (a) GeS and (b) GeSe.

References

- [S1] L. You, Y. Zhang, S. Zhou, A. Chaturvedi, S. A. Morris, F. Liu, L. Chang, D. Ichinose, H. Funakubo, W. Hu, T. Wu, Z. Liu, S. Dong, J. Wang, Origin of giant negative piezoelectricity in a layered van der Waals ferroelectric, *Sci. Adv.* 5(4), eaav3780 (2019).

- [S2] Y. Liu, W. Wang, Z. Wang and C. Si, Ultrahigh Negative Longitudinal Piezoelectricity in Rhombohedral GeTe and Its Group IV–VI Analogues, *Nano Lett.* 25(9), 3630 (2025).
- [S3] F. Xu, H. Su¹, Z. Gong, Y. Wei, H. Jin and H. Guo, Controllable ferroelectricity and bulk photovoltaic effect in elemental group-V monolayers through strain engineering, *Phys. Rev. B* 106(19), 195418 (2022)
- [S4] W. Wang, Y. Xiao, T. Li, X. Lu, N. Xu, Y. Cao, Piezo-photovoltaic Effect in Monolayer 2H-MoS₂, *J. Phys. Chem. Lett.* 15(13), 3549 (2024).
- [S5] Y. Dong, M. M. Yang, M. Yoshii, S. Matsuoka, S. Kitamura, T. Hasegawa, N. Ogawa, T. Morimoto, T. Ideue and Y. Iwasa, Giant bulk piezophotovoltaic effect in 3R-MoS₂, *Nat. Nanotechnol* 18(1032), 36 (2023).