

## REVIEW ARTICLE

# Rational design of photoelectrodes for photoelectrochemical water splitting and CO<sub>2</sub> reduction

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Solar energy has promising potential for building sustainable society. Conversion of solar energy into solar fuels plays a crucial role in overcoming the intermittent nature of the renewable energy source. A photoelectrochemical (PEC) cell that employs semiconductor as photoelectrode to split water into hydrogen or fixing carbon dioxide (CO<sub>2</sub>) into hydrocarbon fuels provides great potential to achieve zero-carbon-emission society. A proper design of these semiconductor photoelectrodes thus directly influences the performance of the PEC cell. In this review, we investigate the strategies that have been put towards the design of efficient photoelectrodes for PEC water splitting and CO<sub>2</sub> reduction in recent years and provide some future design directions toward next-generation PEC cells for water splitting and CO<sub>2</sub> reduction.

**Keywords** photoelectrochemistry, water splitting, CO<sub>2</sub> reduction

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## 1 Introduction

Global warming, as a result of anthropogenic emission of greenhouse gases [e.g., carbon dioxide (CO<sub>2</sub>)], has led to increasing efforts in renewable energy researches and utilizations. Among all the renewable energy sources, solar energy is very attractive as it is the largest renewable source (with ~120 000 TW reaches the Earth's surface) [1, 2]. However, like many other renewable energy sources, solar energy is intermittent and variate diurnally and geographically. Hence, finding efficient and reliable methods to store solar energy is crucial for meeting daily energy demands.

Artificial photosynthesis is a promising and attractive approach to convert solar energy into energy-rich chemical fuels through processes such as water splitting, which generates hydrogen fuel, and CO<sub>2</sub> reduction, which fixes CO<sub>2</sub> into valuable chemicals. Hydrogen fuel can release energy with H<sub>2</sub>O as the only byproduct, which is environmentally benign [1–5]. On the other hand, turning CO<sub>2</sub> into valuable chemicals with solar energy can decrease the atmospheric level of CO<sub>2</sub> and achieve carbon neutrality. There are three general routes available for the artificial photosynthesis: i) Photovoltaic cell combined with electrolysis cell; ii) photocatalytic cell; iii) photoelectrochemical (PEC) cell [2, 6–8]. Each route has its own merits and challenges. In this review, we will mainly focus on PEC cell.

Ever since the pioneering works by Fujishima and colleagues [9, 10], PEC cell has been widely explored in the

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endeavor of finding more efficient water splitting and CO<sub>2</sub> reduction strategies [1, 3, 11–13]. In a PEC cell configuration, the semiconductor is either directly in contact with the electrolyte to form a semiconductor-liquid junction (SCLJ) or separated from the electrolyte by an overlayer with versatile functionalities [13]. Such a cell configuration provides potential cost saving compared with Route 1, as PEC waives the need for an electrolysis cell and thus avoids the fabrication and assemble costs of two-individual cell (photovoltaic cell and electrolysis cell). Besides, unlike a photovoltaic cell which requires p–n junction, PEC cell comes with build-in electric field at the SCLJ. Thus, the SCLJ can drive photo-generated holes (electrons) to the photoanode (photocathode) surface for oxidation (reduction) reaction. Another benefit of PEC cell is the oxidation and reduction products are generated at two photoelectrodes, which can avoid products mixing as it is the case for Route 2. Despite the advantages, many researchers have pointed out that materials requirements for artificial photosynthesis are stringent. As the source of power for artificial photosynthesis is coming from the sunlight that reaches the earth surface ( $\sim 0.1 \text{ W}\cdot\text{cm}^{-2}$  at most), the current density at the PEC electrodes is limited to the order of  $10 \text{ mA}\cdot\text{cm}^{-2}$ . This is far from the output for a commercial electrolyzer ( $\sim 0.5\text{--}2 \text{ A}\cdot\text{cm}^{-2}$ ). This means that a PEC system needs to have an electrode size that is 50–200 times larger than that of a commercial electrolyzer [1, 14]. Thus, the use of cheap, abundant, and stable electrode material for PEC is necessary. Besides, any system with liquid is subjected to the risk of leakage, hence environmentally benign materials for PEC cell are also required.

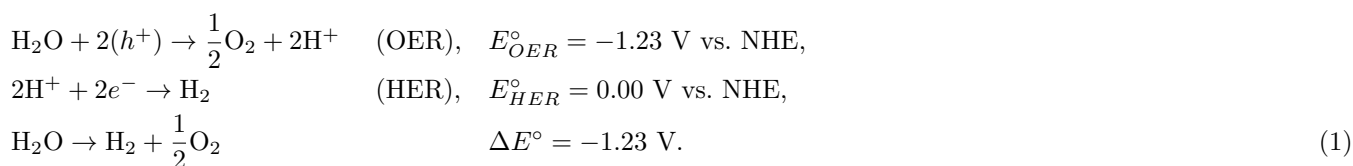
This review focuses on the recent design strategies of PEC photoelectrode. Different strategies have been used

in improving the performance of PEC photoelectrode in water splitting and CO<sub>2</sub> reduction. These strategies include the selection of underlying semiconductor, nanostructuring the semiconductor, applying cocatalyst layer(s) to enrich the semiconductor photocatalytic performance, or doping the semiconductor with different elements. We will first introduce the fundamentals in PEC cell, then review the photoelectrode designs in water splitting and subsequently CO<sub>2</sub> reduction. The intention of the arrangement is to slowly build up the readers' understanding in PEC system through a much simpler water splitting then move forward to CO<sub>2</sub> reduction.

## 2 Fundamentals of PEC in water splitting and CO<sub>2</sub> reduction

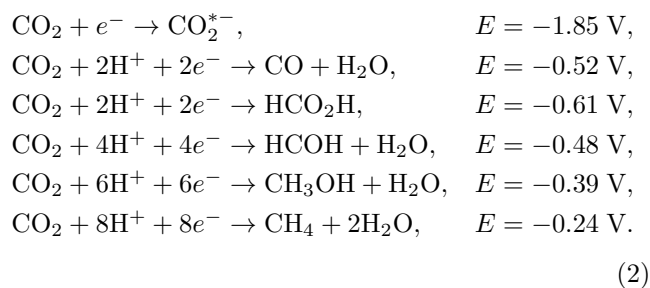
The process of splitting water molecule into hydrogen and oxygen is a thermodynamically uphill process as shown in Eq. (1). The free energy change for the reaction under standard conditions is  $\Delta G = 237.2 \text{ kJ}\cdot\text{mol}^{-1}$ . Using Nernst equation, we can calculate the reversible potential difference of the process to be  $\Delta E^\circ = 1.23 \text{ V}$  per electron transferred. Ideally, a semiconductor requires a bandgap of at least 1.23 eV to drive the water splitting reaction, which can be translated into solar radiation with wavelengths shorter than  $\sim 1000 \text{ nm}$  is needed. If the charge transfer resistance is taken into account, semiconductor with  $\sim 1.6\text{--}2.4 \text{ eV}$  bandgap is required [3].

The overall water splitting reaction is the combination of two half-cell reactions, with one the oxygen evolution reaction (OER) and the other the hydrogen evolution reaction (HER):

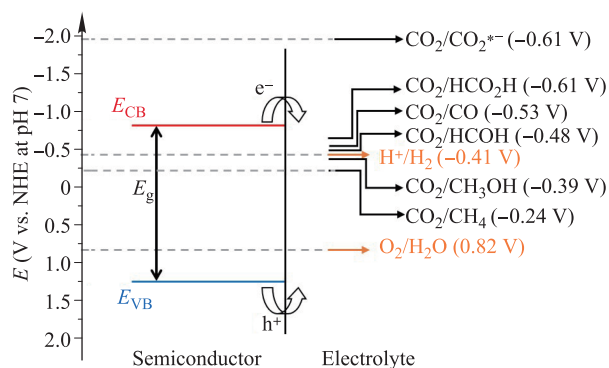


Standard electrode potential  $E^\circ$  (under room temperature and pressure at pH 0) of each of the half reactions can be converted into electrochemical potential, which is given by  $-qE^\circ$ , where  $q$  is the elementary charge. Spontaneous PEC cell reaction requires the conduction band (CB) of the semiconductor to be located more negative than  $E_{\text{HER}}^\circ$  and the valence band (VB) to be located more positive than  $E_{\text{OER}}^\circ$  (shown in Fig. 1).

To reduce CO<sub>2</sub> in a PEC cell, water oxidation reaction is necessary to provide proton (H<sup>+</sup>) to the photocathode to generate various products [as shown in Eqs. (1) and (2)]. The thermodynamic potential of CO<sub>2</sub> reduction to various products with respective to NHE under pH 7 conditions (at room temperature and pressure) are listed in Eq. (2):



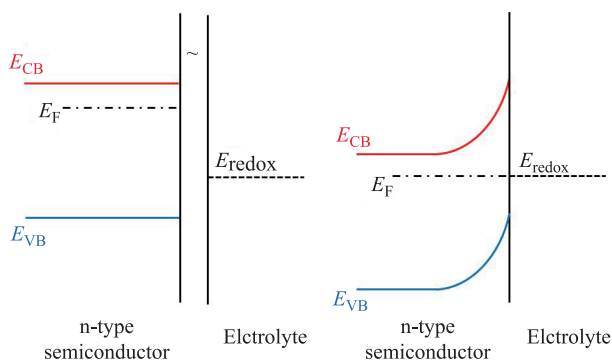
As we can see from these reduction potentials, thermodynamically, the reduction of proton to hydrogen is very close to the reduction of carbon dioxide. Therefore, selectivity of the photoelectrocatalyst used in the process is very important [15]. Note that the difference between the standard reduction potential and the reduction potential



**Fig. 1** Favorable conduction band (CB) and valence band (VB) energy levels for spontaneous oxidation of water and reduction of  $\text{H}^+$  and most of the  $\text{CO}_2$  reactions.

from proton to hydrogen under pH 7 follows Nernst relation:  $E = E^\circ - 0.059x$ , where  $x$  is the value of pH value of the solution. Similar to the band alignment in PEC for water splitting, the conduction band of a photoelectrode for a  $\text{CO}_2$  reduction reaction needs to be lower than the reduction reaction potential (See Fig. 1) and the valence band needs to be higher than the oxidation reaction potential.

Photon energy larger than the bandgap ( $E_g$ ) is required to excite an electron–hole pair in the semiconductor. Separation of the electron–hole pairs before the electron–hole recombination is crucial to improve the efficiency of the photocurrent density from a PEC cell. Unlike a photovoltaic device that requires a p–n junction to drive the separation of electron–hole pairs, PEC device exhibits a built-in electric field at the semiconductor–liquid junction (SCLJ). The built-in electric field is a result of the equilibration of electrochemical potential (Fermi level  $E_F$ ) between the semiconductor and the redox couples of interests in the solution that leave a space charge region within the semiconductor surface. This space charge region also causes the electronic band of the semiconductor to bend (see Fig. 2). Once the electron–hole pairs have been separated, transfer of the charge carriers away from the sur-



**Fig. 2** Band bending at the n-type semiconductor–liquid junction (SCLJ).

face of the semiconductor to the solution to participate in redox reactions is another crucial research topic in PEC cell. Many researches have been focusing on improving the charge separation efficiency and charge injection efficiency through nanostructuring, overlayer film deposition, p–n junction formation and dopant introduction into the semiconductor [3]. Detailed mechanisms of the SCLJ are not shown here. Interested reader can refer to Nozik and Memming’s paper [16].

### 3 Performance characterization

#### • Incident photon to current efficiency (IPCE)

IPCE quantifies the ratio of incident photon that are converted into electrons collected by the PEC cell at different wavelength of the incident light:

$$\begin{aligned} \text{IPCE}(\lambda) &= \frac{\text{collected electrons}}{\text{incident photons}} \\ &= \frac{J_{\text{PEC}}(\lambda)/e}{P_{\text{mono}}(\lambda) \left(\frac{hc}{\lambda}\right)} = \frac{J_{\text{ph}}(\lambda) \times 1239.8}{P_{\text{mono}}(\lambda) \times \lambda}, \end{aligned}$$

where  $J_{\text{PEC}}(\lambda)$  ( $\text{mA}\cdot\text{cm}^{-2}$ ) is the photocurrent density of the PEC cell measured under monochromatic illumination at wavelength  $\lambda$  (nm), with power intensity  $P_{\text{mono}}(\lambda)$  ( $\text{mA}\cdot\text{cm}^{-2}$ ),  $e$  (C) is the charge of one electron,  $h$  (J·s) is Planck’s constant,  $c$  ( $\text{m}\cdot\text{s}^{-1}$ ) is the speed of light, and  $hc/e$  is equal to  $1239.8 \text{ V}\cdot\text{nm}$ .

#### • Absorbed photon-to-current conversion efficiency (APCE)

When we want to know how much of the absorbed photon (instead of incident) is being converted into electron by the PEC cell, a charge generation efficiency term,  $\eta_{e-/h+}$ , i.e., ratio of incident photon that can be utilized by the PEC cell.  $\eta_{e-/h+}$  can be estimated by using UV/Vis spectroscopy.

$$\text{APCE}(\lambda) = \text{IPCE}(\lambda)/\eta_{e-/h+}.$$

### 4 Photoelectrodes for solar water splitting

The design of photoelectrode is crucial to enable the commercialization of PEC water splitting cell. Many factors must be considered into a successful photoelectrode design. A PEC cell often operates under rather corrosive environments, i.e., for photoanode, basic electrolytes were often used, and for photocathode, acidic electrolytes were used instead. It is known that some semiconductor materials can degrade under extreme pH condition. For reliable long-term application, these semiconductor materials must be stable under solar radiation and acid/alkaline conditions.

Transition metal oxides have long been used in solar water splitting research since the pioneering work by Fujishima and colleagues [9], due to their stability and

environment-friendly nature. Popular transition metal oxides used in a PEC water splitting cells include n-TiO<sub>2</sub>, n-WO<sub>3</sub>, n-Fe<sub>2</sub>O<sub>3</sub>, n-BiVO<sub>4</sub>, n-ZnO, p-CuO, p-Cu<sub>2</sub>O, and p-NiO. Each material has its shortcomings and these shortcomings have been addressed in several reports. In this section, we will review the strategies that have been used in recent years to improve the PEC performance and stability of a few selected transition metal oxides.

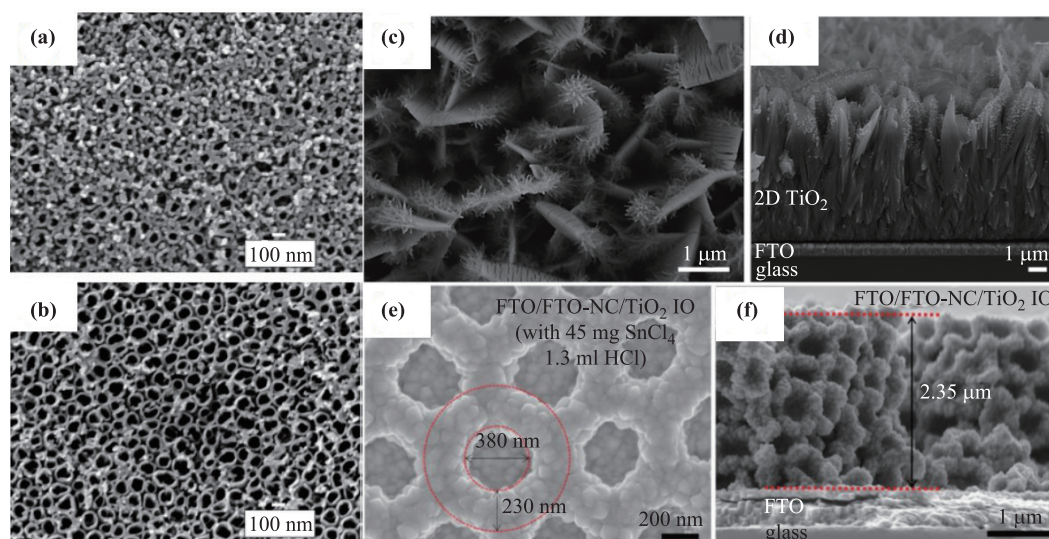
## 4.1 Photoanodes

### 4.1.1 TiO<sub>2</sub> photoanode

Titanium dioxide (TiO<sub>2</sub>), an n-type semiconductor with a bandgap  $\sim 3.2$  eV [17–20], is widely used in PEC water splitting research since Fujishima's work in 1972 [9], mainly due to its high photostability, non-toxicity, abundance, and low cost. However, the wide bandgap limits its ability to utilize visible light and the low charge carrier mobility will induce electron–hole recombination.

Nanostructuring TiO<sub>2</sub> photoelectrode to increase the surface-to-volume ratio has been a popular method to enhance the PEC performance. One-dimensional (1D) nanostructures have drawn many attentions due to the potential ability to provide fast electron transport pathway along the longitudinal direction. Anodized TiO<sub>2</sub> nanotubes [Figs. 3(a) and (b)] were reported by Grimes's group for solar water splitting [21, 22]. The length, the wall thickness, and the spacing of the highly ordered TiO<sub>2</sub> nanotube arrays grew on Ti foils can be adjusted by tuning the applied potential and chemical bath temperature. The optimized TiO<sub>2</sub> nanotube was able to achieve 6.8% photoconversion efficiency. Kim *et al.* demonstrated that

the performance of the nanotubes can be improved by growing nano-branch layer on the side-walls of the nanotubes [23]. The nano-branch was a result of TiCl<sub>3</sub> solution treatment at 80 °C on the surface of TiO<sub>2</sub> nanotubes. The surface treatment was also shown to create oxygen vacancies on the surface of the nanotubes, which can enhance the charge transport efficiency by increasing the donor concentration. The nano-branch layer enhances the charge transfer efficiency by providing additional active sites for water adsorption. Besides 1D nanostructure, two-dimensional (2D) nanostructures have also shown promising results. Butburee *et al.* synthesized a 2D anatase TiO<sub>2</sub> porous single-crystalline nanostructure (PSNs) on FTO for PEC water splitting through an ion exchange-induced pore-forming process [Figs. 3(c) and (d)] [18]. The single crystalline nature of the nanostructure allows long-range charge diffusion length and structural coherence; the porous nature of the nanostructure allows abundant active sites for chemical reaction. The TiO<sub>2</sub> PSNs exhibit very low onset potential and a photocurrent density of 1.02 mA·cm<sup>-2</sup> at 1.23 V vs. RHE (denoted as V<sub>RHE</sub> hereafter), which is close to the theoretical limit of TiO<sub>2</sub> (1.12 mA·cm<sup>-2</sup>). One other popular strategy to improve the surface area is “host and guest” composite photoelectrode [19, 20]. Wang *et al.* constructed a 3D FTO inverse opal nanocrystal as conductive host for atomic layer deposited TiO<sub>2</sub> using polystyrene opals [Figs. 3(e) and (f)] [19]. The highly porous host provides a shorter pathway for photo-generated charge carriers from the TiO<sub>2</sub> photoelectrode. This work demonstrated a way to improve PEC performance through highly porous skeleton without compromising the electron collection. The TiO<sub>2</sub> on the inverse opal nanocrystal was able to achieve photocurrent density



**Fig. 3** Different TiO<sub>2</sub> morphology through nanostructuring: (a, b) Anodized TiO<sub>2</sub> nanotube. Reprinted with permission from Ref. [22]. (c, d) 2D anatase TiO<sub>2</sub> porous single-crystalline nanostructure (PSNs). Reproduced with permission from Ref. [18]. (e, f) 3D FTO inverse opal nanocrystal as conductive host for atomic layer deposited TiO<sub>2</sub>. Reproduced with permission from Ref. [19].

of  $1.0 \text{ mA}\cdot\text{cm}^{-2}$  at  $1.23 V_{\text{RHE}}$ .

Due to the wide bandgap nature,  $\text{TiO}_2$  is often coupled with another semiconductor with smaller bandgap to improve the amount of solar energy being harvested by the photoelectrode. Liu *et al.* deposited bismuth vanadate ( $\text{BiVO}_4$ ), an n-type semiconductor with bandgap  $\sim 2.4 \text{ eV}$ , onto inverse opal nanostructured  $\text{TiO}_2$  to enable the photoelectrode to absorb wider range of solar spectrum [20]. The band structure of these two semiconductors also allow the formation of the favorable type 2 heterojunction, which can enhance the charge separation within the semiconductor. Further, they used photo-assisted electrodeposition to deposit cobalt-phosphate (Co-Pi) on  $\text{BiVO}_4$  that leads to 9.0 times photo-to-energy conversion efficiency improvement over  $\text{TiO}_2$  inverse opal. Another work created a hierarchical  $\text{Fe}_2\text{O}_3$  (bandgap  $\sim 2.2 \text{ eV}$ ) nanorods with  $\text{TiO}_2$  nanosheets to enhance visible light absorption [24].

Introducing foreign elements is another popular strategy to improve the performance of  $\text{TiO}_2$  through improving the visible light absorption and charge carrier mobility. Many elements have been used as dopant in  $\text{TiO}_2$  photoanode, including but not limited to Fe, Rh, Co, Ni, and Si for cations; N, C, S, and B for anions [25–27]. However, recent interests have been attracted by doping  $\text{TiO}_2$  with two different elements due to their advantages over monodoped  $\text{TiO}_2$  in higher photocatalytic activity and visible light absorption capability. The C/N codoped  $\text{TiO}_2$  NW as reported by Song *et al.*, was able to achieve a 253-fold improvement in visible light photoactivity when compared to bare  $\text{TiO}_2$  NW, which is among the best reported PEC efficiencies under visible irradiation. The synergistic effect of the codopants was to improve visible light absorption capability and charge separation and transference efficiencies.

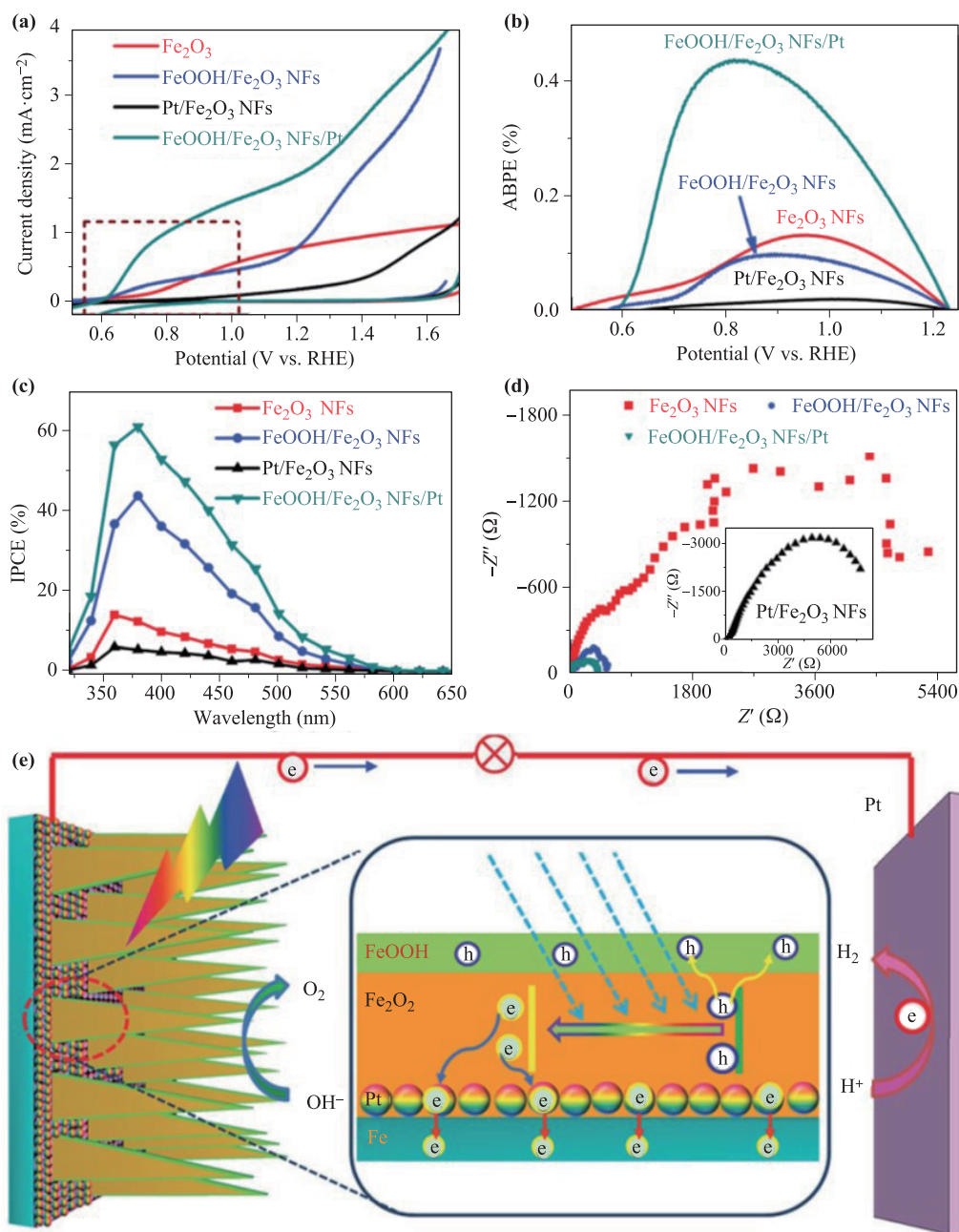
#### 4.1.2 $\alpha\text{-Fe}_2\text{O}_3$ photoanode

Hematite ( $\alpha\text{-Fe}_2\text{O}_3$ ) for solar water splitting has been researched extensively since Hardee and Bard discovered its PEC property in 1976 [28]. Hematite has many attractive properties as a promising photoelectrochemical solar water splitting material, in particular it has a bandgap of 1.9 eV to 2.2 eV that can harvest majority of the visible solar spectrum, which corresponds to a maximum photocurrent density of  $12.6 \text{ mA}\cdot\text{cm}^{-2}$  [11]. It has a conduction band that is more positive than the HER potential of water therefore a potential bias is needed to drive the reaction [29, 30]. It is stable in neutral and alkaline aqueous solution under photocorrosion. Iron is an abundant element in the earth crust and iron oxide is an environmentally friendly material. These characteristics of hematite make it an extremely promising material for solar water splitting. However, its performance still falls behind expectation due to bulk and surface charge recombinations [31], short hole diffusion length (2–4 nm) [32], high overpotential for measurable photocurrent [11], long photon absorp-

tion depth [33, 34], and improper flat band position ( $0.4 V_{\text{RHE}}$ ).

To address these challenges, approaches such as nanostructuring the hematite photoanode into nanotubes [35], nanowire [36], or nanoflakes [37], applying cocatalyst layer, and introducing foreign atom have been taken. Typically, combinations of these approaches are required to yield higher PEC performance. Most notably, Gratzel and co-workers reported a Si-doped cauliflower-like hematite modified with  $\text{IrO}_2$  cocatalyst to achieve a photocurrent density of  $3.75 \text{ mA}\cdot\text{cm}^{-2}$  [38]. This PEC performance from hematite was a benchmark for some time until a record photocurrent density of  $4.32 \text{ mA}\cdot\text{cm}^{-2}$  was reported on a wormlike hematite modified with Pt and Co-Pi [39].

In recent years, layered double hydroxide (LDH) type cocatalysts have been popular due to their high OER catalytic activities. Huang *et al.* deposited NiFe LDH on Mn-doped hematite photoanode for solar water splitting [40]. Mn doping improved charge separation efficiency, while NiFe LDH shifts the onset potential negatively and enhances the charge injection efficiency to 90% at high potentials. The influence of the thickness of the coating was investigated. It was found that although thicker NiFe LDH improves light absorbance, the PEC performance is lower compared to thin overlayer. This is because thick NiFe LDH affects the light absorption on hematite. IPCE of the optimum photoanode was 25% at 350 nm wavelength incident light and  $1.23 V_{\text{RHE}}$ . LDH type cocatalysts were also used in other works, such as NiFeAl LDH [41], ZnCo LDH [42], and NiFe LDH [35]. In another work, dual cocatalyst layers were used. Wang *et al.* fabricated  $\text{Ni}(\text{OH})_2/\text{IrO}_2$  cocatalyst coated on Ti-doped  $\text{Fe}_2\text{O}_3$  photoanode [43]. The synergetic effect of the cocatalyst was investigated. Based on CV test and transient photocurrent test,  $\text{Ni}(\text{OH})_2$  cocatalyst was shown to be able to store charges (holes) from the photoanode, which has been widely reported [44].  $\text{IrO}_2$  on  $\text{Ni}(\text{OH})_2$  is said to further improve the OER efficiency.  $\text{IrO}_2$  cocatalyst alone does not improve the charge injection efficiency much while  $\text{Ni}(\text{OH})_2$  alone improve the charge injection efficiency a little at low applied potential. However, with both catalysts loaded, the PEC performance was significantly improved. Significant increase in photovoltage can be observed with  $\text{Ni}(\text{OH})_2$  loading, leading to negative shift of the onset potential. The optimum photoanode has an IPCE of  $\sim 34\%$  at 350 nm wavelength incident light at  $1.23 V_{\text{RHE}}$ . Wang *et al.* synthesized 1D hematite nanoflakes with the modification of Pt and FeOOH ( $\text{FeOOH}/\text{Fe}_2\text{O}_3$  NFs/Pt) through a single step immersion of hematite nanoflakes in 5 mM  $\text{H}_2\text{PtCl}_6$  solution. Interestingly, the Pt nanolayer grew only on the iron foil, while the ultrathin FeOOH nanolayer form on the surface of hematite nanoflakes [see Fig. 4(e)]. This selectively integrated hole-transfer layer (FeOOH) and electron collector layer (Pt) into hematite nanoflakes photoanode can reduce the charge recombination at the interface, which improved photocurrent density at low applied bias



**Fig. 4** (a) Photocurrent density vs. potential ( $J$ - $V$ ) curves of different samples. (b) Applied bias photo-to-current efficiency (ABPE) curves of different samples. (c) IPCE measurements of different samples. (d) Electrochemical impedance spectroscopy (EIS) measurements of different samples. (e) Schematic illustration of charge separation and transfer on FeOOH/Fe<sub>2</sub>O<sub>3</sub> NFs/Pt. Reproduced with permission from Ref. [45].

[shown in Fig. 4(a)]. IPCE has been drastically improved and the Nyquist plot measurement indicated the reduction of charge transfer resistance upon the modification [see Figs. 4(c) and (d)].

The effect of different dopants has on the function of cocatalyst layer plays on the hematite photoanode was investigated by Tsygonok [46]. It was found that with Sn doping, Fe<sub>1-x</sub>Ni<sub>x</sub>OOH cocatalyst improves the PEC performance by reducing the recombination current. On the other hand, for Zn-doped hematite, Fe<sub>1-x</sub>Ni<sub>x</sub>OOH cocat-

alyst not only reduces recombination current, but also improves hole current. This study suggested a possible link between the surface and bulk PEC processes.

#### 4.1.3 BiVO<sub>4</sub> photoanode

Bismuth Vanadate (BiVO<sub>4</sub>) is another promising n-type semiconductor material with a bandgap slightly larger than hematite at 2.4 eV, comprises of nontoxic, low cost material with good stability. The valence band edge of

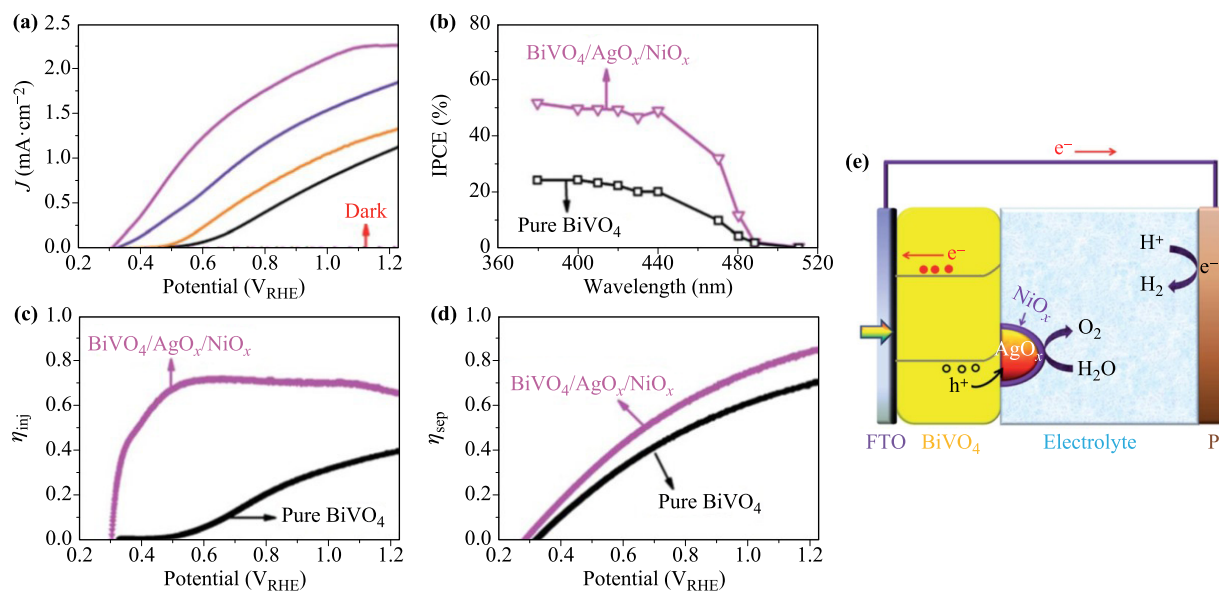
bismuth vanadate is approximately at 2.4 eV<sub>RHE</sub> which means it has a conduction band minimum closed to 0 eV (0.02 eV<sub>RHE</sub>) [47], hydrogen reduction potential at standard condition, the onset potential for photoelectrochemical measurement should be closed to 0 V<sub>RHE</sub>. Under standard AM 1.5 G solar illumination, the theoretical STH efficiency of this material is 9.2%, an efficiency closed to the commercialized requirement for solar water splitting [48]. Despite the beneficial characteristics, BiVO<sub>4</sub> suffers from some drawbacks that lead to its low performance. Major challenges of BiVO<sub>4</sub> are: i) Poor OER kinetics, which results in surface recombination. ii) Short minority carrier diffusion length, ~70 nm. iii) Less than optimal CB edge position which requires an external bias to drive the thermodynamics of water splitting.

To overcome the aforementioned challenges, a combination of photoelectrode design strategies are often used. Lv *et al.* fabricated BiVO<sub>4</sub> porous photoanodes modified with carbon dots (CD) and NiFe-LDH to enable efficient solar water splitting [49]. The improvement of PEC performance is mainly attributed to the catalytic ability arises from the synergetic effect between CD and NiFe-LDH. The photocurrent density improved from 0.79 mA·cm<sup>-2</sup> for the bare BiVO<sub>4</sub> to 2.84 mA·cm<sup>-2</sup> for CDs/NiFe-LDH/BiVO<sub>4</sub>. The IPCE values improved from 19.17% to 40.94% at 380 nm at 1.23 V<sub>RHE</sub>. In another work, Xu *et al.* modified BiVO<sub>4</sub> photoanode by two cocatalysts, AgO<sub>x</sub> and NiO<sub>x</sub>, to improve the charge injection efficiency and the bulk charge separation efficiency [50]. EIS showed significant reduction of surface charge transfer resistance of the BiVO<sub>4</sub>/AgO<sub>x</sub>/NiO<sub>x</sub> sample at the interface with electrolyte. The use of photoassisted kelvin probe force mi-

croscopy (KPFM) showed that the final sample has higher surface photovoltage (SPV) compared to other samples that can facilitate large band bending for efficient bulk charge separation. IPCE was measured to be ~49.3% in the range of 380-450 nm wavelength at 1.23 V<sub>RHE</sub> [see Figs. 5(a)–(d)]. Schematic illustration of the photoelectrode design and solar water splitting mechanism is shown in Fig. 5(e).

Prussian blue type catalysts have also been used recently, as they are cheap, environmentally friendly, highly stable, and easy to synthesize [51, 52]. Recently, Shaddad *et al.* fabricated Zr-doped BiVO<sub>4</sub>/Nickel Iron Prussian blue (NiFePB) core shell photoanode [53], which were found to have improved charge transfer and charge separation efficiency over bare Zr-doped BiVO<sub>4</sub>. EIS test showed a decrease in charge transfer resistance, especially at low potential. Bandgap was shown to decrease with the deposition of the NiFePB. High stability was achieved for a 50 h stability test under simulated solar irradiation.

Tandem cell with two semiconductors forming a heterogeneous junction can be helpful in facilitating charge transport within the semiconductors. For instance, Kim and Choi constructed a heterojunction of BiVO<sub>4</sub>/ZnFe<sub>2</sub>O<sub>4</sub> that was able to achieve ~2 mA·cm<sup>-2</sup> at 1.23 V<sub>RHE</sub> [54]. The ZnFe<sub>2</sub>O<sub>4</sub> layer also improved the stability of BiVO<sub>4</sub> photoanode. The design of two heterojunctions has also been reported. BiVO<sub>4</sub>/WO<sub>3</sub>/SnO<sub>2</sub> photoanode was able to achieved ~3.1 mA·cm<sup>-2</sup> at 1.23 V<sub>RHE</sub> with IPCE value at ~70% at 300 nm wavelength light at 1.23 V<sub>RHE</sub> [55]. In another work, McDowell deposited TiO<sub>2</sub> and Ni thin layer onto BiVO<sub>4</sub> to achieve stable photocurrent over a duration of 4 hours [56].



**Fig. 5** (a)  $J$ - $V$  curves with BiVO<sub>4</sub>/AgO<sub>x</sub> (orange), BiVO<sub>4</sub>/NiO<sub>x</sub> (violet). (b) IPCE curves. (c) Charge injection efficiency. (d) Charge separation efficiency. (e) Schematic of solar water splitting mechanisms on BiVO<sub>4</sub>/AgO<sub>x</sub>/NiO<sub>x</sub>. Reproduced with permission from Ref. [50].

## 4.2 Photocathode

### 4.2.1 p-NiO photocathode

p-NiO is a p-type semiconductor with reported bandgap ranging from 3.3 eV [57] to 4.3 eV [58]. The Fermi level is  $\sim 0.5$  eV<sub>RHE</sub> (at pH 0) with valence band maximum (VBM) at  $\sim 0.4$  eV more positive than the Fermi level (i.e., VBM is  $\sim 0.9$  eV<sub>RHE</sub>) [58–61]. NiO is an insulator at room temperature [62]. However, its conductivity can be improved by increasing the concentration of Ni<sup>3+</sup> through lithium doping, nickel vacancies, or interstitial oxygen in p-NiO crystallites [63–65]. The wide bandgap of p-NiO semiconductor limits its ability to utilize wider solar spectrum, thus limit the maximum photocurrent density that can be generated from the semiconductor. As a result, there are relatively limited works on using p-NiO for PEC water splitting.

In an effort of nanostructuring the p-NiO photocathode, nickel foil was first anodized under different applied voltages and subsequently annealed under different temperature to get the desirable porosity and surface properties [66–68]. Commonly, the morphology results from anodization is nanotubes for other metals, however, macroporous structure was obtained for NiO due to rapid dissolution of the oxidized product. Hu *et al.* reported using anodized p-NiO microporous structure for PEC water splitting. Al<sub>2</sub>O<sub>3</sub> was used to modify the surface of NiO to suppress the back recombination between photogenerated electron–hole pairs and stabilize the p-NiO photocathode. Sapi *et al.* used Pt nanoparticles as cocatalyst to further improve the p-NiO film. The changes in charge transfer efficiency and porosity upon the Pt nanoparticles loadings were investigated [68]. Flake-like nanostructured p-NiO can be obtained through a hydrothermal method [69]. WO<sub>3</sub> photoanode with smaller bandgap ( $\sim 2.26$  eV) was deposited on the p-NiO nanoflake to widen illumination absorption range. The p–n junction can facilitate separation of photo-generated electron–hole pairs, thus improved the photocurrent density by 3.94 times compared to p-NiO nanoflake.

Introducing a lower bandgap semiconductor to p-NiO can widen the absorption range on solar energy spectrum. Suzuki *et al.* deposited CdS (bandgap  $\sim 2.3$  eV) nanoparticles on p-NiO as photosensitizer, which greatly enhanced the visible light absorption [67]. In another work, CdSe (bandgap  $\sim 1.74$  eV) was used as photosensitizer for p-NiO [70]. Introducing organic dye onto p-NiO photocathode has been a popular practice in dye sensitize solar cell research [71]. There are a few works that utilized dye sensitized p-NiO in PEC hydrogen generation. Excitation of a dye molecule by solar energy can promote an electron from the highest occupied molecular orbital (HOMO) to the lowest unoccupied molecular level (LUMO). In this case, the HOMO–LUMO gap of a dye molecule is analogous to the band gap of a semicon-

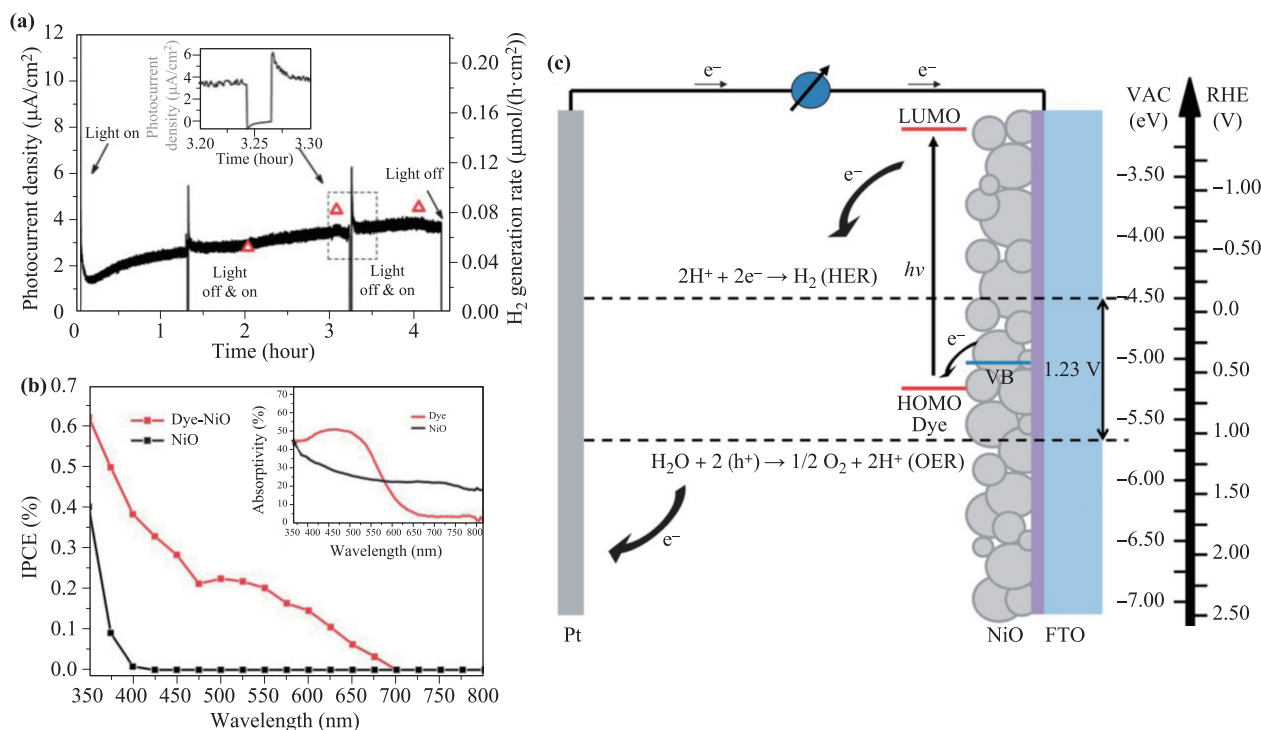
ductor. Unlike dye-sensitized photoanodes, dye-sensitized NiO photocathodes have shown to be stable under alkaline and neutral aqueous environment for HER [72, 73]. Tong *et al.* achieved stable HER with a monolayer of PMI-6T-TPA donor-acceptor type dye deposited on p-NiO photocathode [Fig. 6(a)]. The incorporation of dye improved visible light absorption of p-NiO and facilitated charge transfer from the photocathode to the electrolyte [Fig. 6(b)]. A schematic diagram of the dye-sensitized p-NiO is shown in Fig. 6(c). Some researchers even demonstrated its stability under acidic solution [74]. More detailed and comprehensive review on dye sensitized photocathode has been conducted previously [74].

## 5 Photoelectrochemical (PEC) CO<sub>2</sub> reduction systems

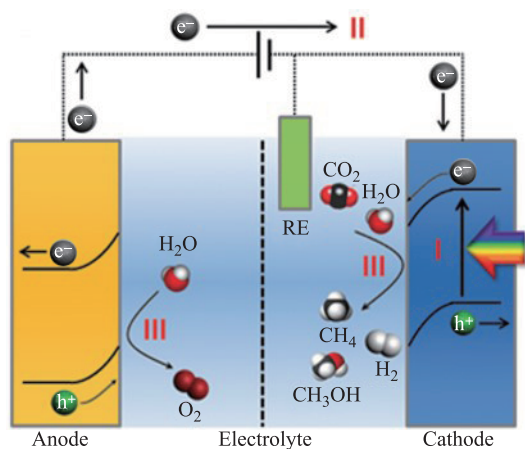
Generally, the PEC cell for CO<sub>2</sub> reduction consists of a semiconductor photocathode and a counter anode (Fig. 7). The photocathode absorbs light to produce electron–hole for catalytic reactions. The light utilization efficiency, including the absorption of incident photons, charge separation and migration, is pivotal for the performance of the PEC cell. However, PEC CO<sub>2</sub> reduction system usually suffers from weak light harvesting, serious charge recombination and slow electron migration rate at photocathodes [75]. Engineering the nanostructure or electronic properties of photocathode has been shown as effective methods to overcome the above drawbacks. In this section, we will mainly summarize the popular designs of photocathodes for improving the efficiency of CO<sub>2</sub> reduction.

### 5.1 Introduction of hetero-dopants

Introduction of dopant is an effective way for modifying the electronic properties of semiconductor and tuning the light harvesting efficiency [77]. The dopants can alter the type of semiconductor. A representative example is that the doped silicon [78]. Introducing the N or P element into silicon results in n-type properties, while B or Ga element results in p-type properties. In addition, dopant introduction can alter the CB or VB position of semiconductor. Non-metal dopant usually affects the CB edge but hybridizes VB to shift it upward, and thus narrows the band gap [79]. However, metal-based dopants can induce the formation of impurity in forbidden band and then increase the charge transfer ability by serving as either an electron donor or an acceptor. The band gap can be narrowed or splitted into a two-step excitation process by introducing impurity, leading to an obvious redshift of light absorption [80]. For example, the carrier density in Nb-doped TiO<sub>2</sub> nanowires was up to  $\sim 10^{21}$  cm<sup>-3</sup>, which is a dramatic improvement compared to the original sample [81]. In addition, the metal dopants can also act as active centers for catalysis reactions. Such as the Cu-doped TiO<sub>2</sub>, which



**Fig. 6** (a) Chronoamperogram (left axis) and hydrogen generation rate (right axis, red triangles) of dye-sensitized p-NiO. (b) IPCE (%) spectrum and absorptivity (inset) UV-Vis absorbance of p-NiO and dye-sensitized p-NiO. (c) Schematic of charge transfer for HER on the dye-sensitized p-NiO photocathode. Reproduced with permission from Ref. [73].



**Fig. 7** Schematic of PEC  $\text{CO}_2$  electrolyzer with a photocathode for  $\text{CO}_2$  reduction, a photoanode for OER, and a reference electrode. Reproduced with permission from Ref. [76].

shows good selectivity in PEC  $\text{CO}_2$  conversion to  $\text{CH}_3\text{OH}$  [82]. Li *et al.* reported the N-doped CuO with more negative CB and lower overpotential for high efficiency PEC conversion of  $\text{CO}_2$  to methanol [83]. The carrier concentration of N-doped CuO reaches  $7.5 \times 10^{11} \text{ cm}^{-3}$ , which is about  $10^8$  times of that in pristine CuO film. In 2016, Ohno *et al.* incorporated boron into graphitic carbon nitride ( $g\text{-C}_3\text{N}_4$ ) and realized a 5-time improvement of the PEC current compared to the pure  $g\text{-C}_3\text{N}_4$  [84].

Despite the doping strategy has been widely investigated on the design of photocathodes, its development for PEC  $\text{CO}_2$  reduction is still in infant stage. We expect that more types of dopants can be discovered for constructing novel photocathodes and improving the efficiency of PEC  $\text{CO}_2$  reduction system.

## 5.2 One-dimensional nanostructures

Benefiting from the unique morphology and electronic characteristics, one-dimensional (1D) nanostructures have been widely used in high-performance PEC water splitting and shows immense potentials in PEC  $\text{CO}_2$  reduction [85, 86]. For 1D nanomaterials, the electronic wavefunction is confined within the nanoscale directions, which leads to the quantum confinement effect and then benefits the light utilization [87]. Moreover, its high length-to-diameter ratio can improve light absorption and shorten the length of electron migration, which finally results in high photon-to-electron conversion efficiency [88].

In recent, some 1D semiconductor-based photocathode have been reported for PEC  $\text{CO}_2$  reduction. Taking the 1D p-Si as an example, the faradaic efficiency was doubled with 1D p-Si photocathode, compared to the planar counterpart [89]. Another typical case is the construction of Cu-ZnO/GaN/ $n^+ \text{-p}$  Si photocathode into nanowire arrays [Figs. 8(a) and (b)] [90], which realized the tunable syngas ( $\text{CO}/\text{H}_2$ ) production from  $\text{CO}_2$  and  $\text{H}_2\text{O}$  with a 70% faradaic efficiency for CO at overpotential of 180 mV. In

2016, Yang *et al.* utilized the semiconductor Si nanowires as the substrate and directly assembled Au<sub>3</sub>Cu nanoparticle on its surface for PEC CO<sub>2</sub> reduction [Figs. 8(c) and (d)] [91]. This rationally designed photocathode showed a high CO<sub>2</sub>-to-CO selectivity of ~80% at -0.20 V<sub>RHE</sub> with suppressed HER. The enhancement could be partially ascribed to the superior light trapping and efficient charge transfer induced by the 1D structure. The 1D structure favors photon harvesting and exposes more active sites, exhibiting high selectivity and faradaic efficiency for PEC CO<sub>2</sub> conversion. Similar improvements toward PEC CO<sub>2</sub> reduction have been achieved on other 1D nanostructures based on p-type metal oxides, such as Cu<sub>2</sub>O [92, 93] and Co<sub>3</sub>O<sub>4</sub> [94], etc.

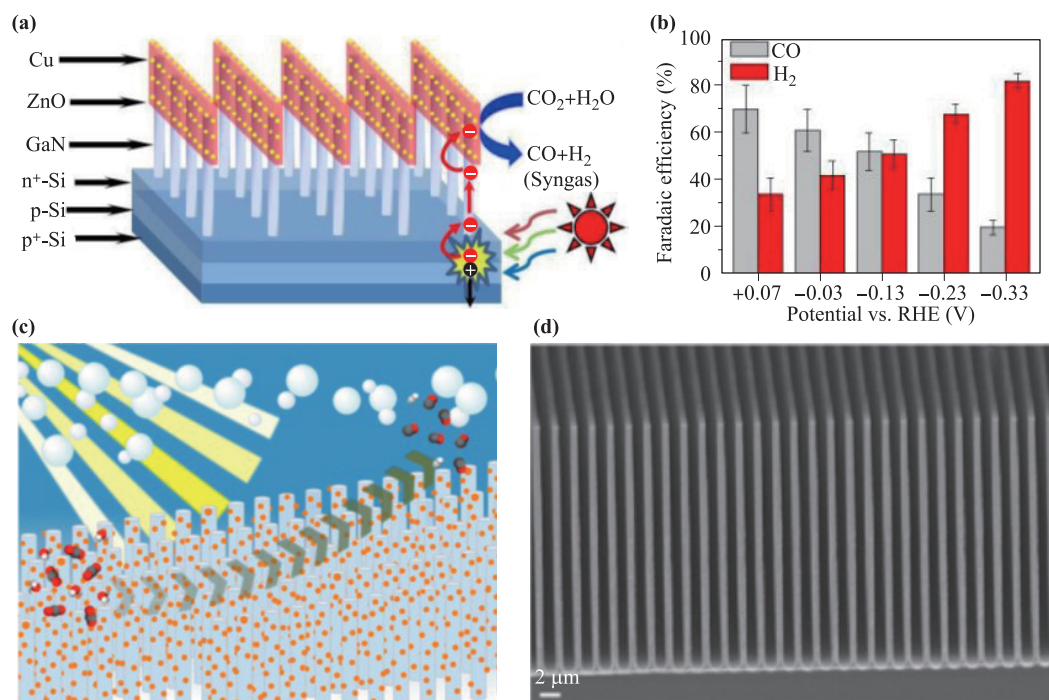
### 5.3 Construction of heterojunctions

It is difficult to simultaneously optimize the three processes of light harvesting, charge separation and migration, and redox reaction in a single material owing to the thermodynamic limitation in PEC systems. However, the barriers could be overcome by integrating semiconductors with Schottky junctions. The Schottky junction can promote the migration of charge carriers (i.e., electrons for p-type semiconductors), and then increase the separation and utilization of the electrons [95]. For example, Lee *et al.* reported that the Schottky junction by anchoring Au cocatalyst, which has smaller work function of 5.1-5.3 eV, on ZnTe (5.79 eV) can significantly improve

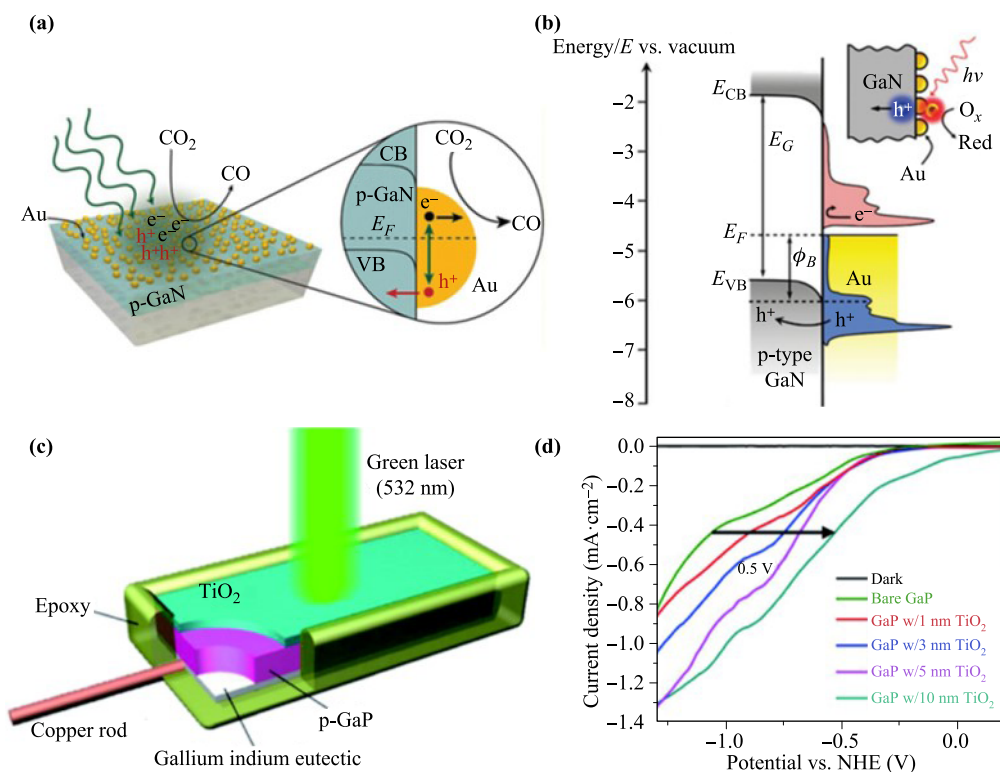
the conversion efficiency of the ZnTe photocathode. This is because the Schottky junction (Au/ZnTe) enhanced the band bending of ZnTe and then promoted the migration of electrons from the bulk to Au, and finally transferred them to the electron acceptor in electrolyte. Moreover, the surface Au cocatalyst can provide reaction sites to CO<sub>2</sub> reduction by suppressing the undesired hydrogen evolution reaction [96]. Recently, Atwater *et al.* reported an Au/p-GaN Schottky junction by loading the Au nanoparticles on p-GaN photocathode, which promoted the transfer of photogenerated electrons from the p-GaN to Au and the hot holes in the opposite direction [Figs. 9(a) and (b)] [97].

Besides ZnTe, Cu<sub>2</sub>O is also a promising semiconductor for PEC CO<sub>2</sub> reduction. The high-efficiency charge-separation effect of the Schottky junction between Au-Cu nanoalloys and a 3D graphene/Cu<sub>2</sub>O photocathode has been demonstrated by Hou *et al.* Such a construction showed effective conversion of CO<sub>2</sub> to methanol in PEC system [98].

Although the p-type photocathodes are widely used in PEC system, they suffer from relatively low stability in aqueous solutions [100]. Coating the photocathode with a heterojunction layer has been demonstrated as an effective approach to improve its corrosion resistance. In this design, the minimal band offset and high-quality interface are needed for facilitating charge transport and suppressing recombination respectively [101]. Silicide [102], TiO<sub>2</sub> [103] and SrTiO<sub>3</sub> [104] have been demonstrated as protective layers in PEC CO<sub>2</sub> reduction. For example, as shown



**Fig. 8** (a) Schematic of 1D Cu-ZnO/GaN/n<sup>+</sup>-p Si photocathode. (b) Faradaic efficiencies of CO and H<sub>2</sub> on 1D Cu-ZnO/GaN/n<sup>+</sup>-p Si photocathode in 0.5 M KHCO<sub>3</sub> electrolyte. Reproduced with permission from Ref. [90]. (c, d) Schematic and SEM image of the 1D Si nanowires with Au<sub>3</sub>Cu nanoparticle cocatalyst. Reproduced with permission from Ref. [91].



**Fig. 9** (a) Schematic illustration of Au nanoparticle as cocatalyst anchored on p-GaN photocathode for PEC reduction of CO<sub>2</sub> to CO. (b) Energy band scheme of a plasmonic Au metal in contact with p-GaN and the corresponding Schottky junction. Reproduced with permission from Ref. [97]. (c) Schematic for the p-GaP photocathode passivated by TiO<sub>2</sub> layer. (d) Photocatalytic plots of p-GaP photocathodes coated TiO<sub>2</sub> layer at various thicknesses. Reproduced with permission from Ref. [99].

in Fig. 9(c), a n-type TiO<sub>2</sub> layer with a thickness of 10 nm was coated on p-GaP using atomic layer deposition (ALD) [99], which passivated the surface and provided a substantial enhancement on the photoconversion efficiency. Moreover, a p–n junction was formed in this construction. Benefiting from the above merits, the heterojunction photocathode showed a 0.5 V reduction in onset overpotential for CO<sub>2</sub> to CH<sub>3</sub>OH [Fig. 9(d)].

#### 5.4 Utilizations of cocatalysts on photocathodes for PEC CO<sub>2</sub> reduction

The efficiency of CO<sub>2</sub> reduction is limited by the sluggish kinetics, which leads to a large overpotential and decreases the conversion efficiency. The CO<sub>2</sub> adsorption and activation are the initial steps for the subsequent PEC reductions, in which the photocathodes act as both light harvesting antennas and catalytic centers. Despite huge efforts have been devoted to improving light utilization efficiency, photocathode surface shows poor adsorption and activation abilities for CO<sub>2</sub>, leading to low conversion efficiency of CO<sub>2</sub> to hydrocarbon [105, 106]. Anchoring cocatalysts on semiconductor surface has been proved to be an effective approach to improve the efficiency of PEC CO<sub>2</sub> reduction [107–109]. The cocatalysts can lower the over-

potential and reduce the energy barrier, facilitating CO<sub>2</sub> activation as well as improving the selectivity. In this part, we will introduce various types of cocatalysts and discuss their roles in the CO<sub>2</sub> reduction process.

##### 5.4.1 Metal-based cocatalysts

Integrating semiconductor with metal nanoparticle cocatalysts is an effective way to improve the PEC performance [109, 110]. The metal cocatalysts can trap the photo-induced charges to promote charge separation and provide active centers to facilitate CO<sub>2</sub> reduction reactions. Besides, the undesired side reaction of HER can be suppressed in the PEC CO<sub>2</sub> reduction system with the right metal cocatalyst(s).

Some metals are competent to act as active sites for the reduction of CO<sub>2</sub> to CO, such as Au [111], Ag [112], and Pd [113], while some other metals such as Sn [114] and Pb [115] show good selectivity for producing formate from CO<sub>2</sub>. Moreover, Cu can produce various hydrocarbon products when used as electrode for CO<sub>2</sub> reduction due to the poor selectivity. Nevertheless, Cu has received extensive attention [98, 116, 117]. Kaneco *et al.* [118] anchored the Pb, Ag, Au, Pd, Cu, and Ni on p-InP semiconductor for PEC CO<sub>2</sub> reduction, and reported that the Pb, Ag, Au and Cu can produce both CO and HCOOH

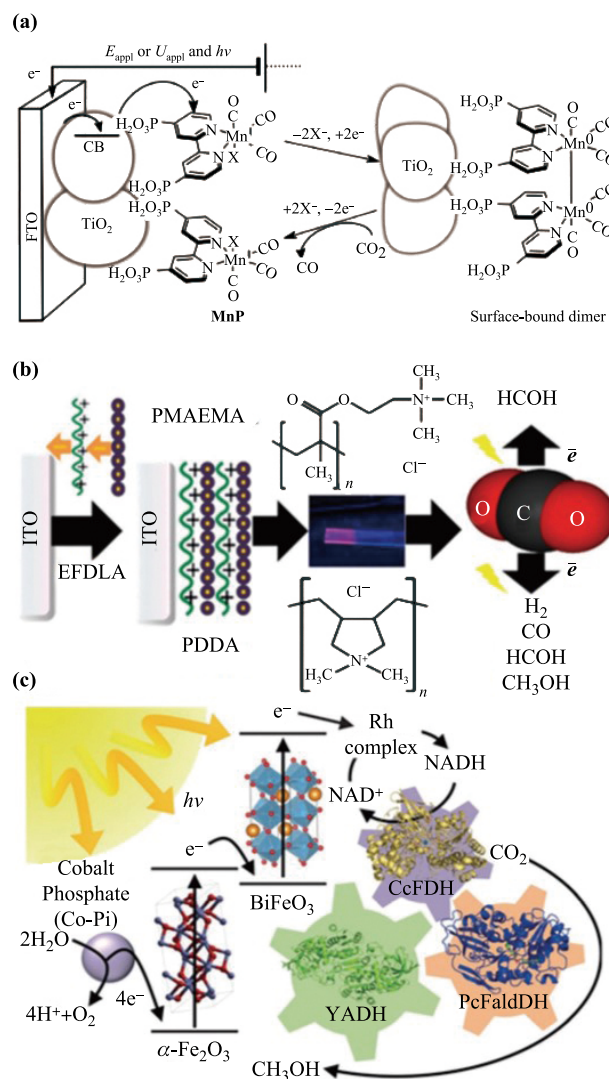
and Pd shows good selectivity for CO. By summarizing the previous reported literature, it can be concluded that anchoring metal nanoparticles on photocathode can positively shift the overpotential of CO<sub>2</sub> reduction and control the selectivity of products. For instance, by modifying the p-Si photocathode with Cu, Ag, and Au for PEC CO<sub>2</sub> reduction, the overpotential was positively shifted and the major products also changed from CO and HCOOH to CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub> [119].

Besides the noble metal cocatalysts described above, several earth-abundant metals have been demonstrated as effective cocatalysts for PEC CO<sub>2</sub> reduction, such as Co [123], Ni [124], Mn [125], and Fe [126]. For example, a highly active Fe porphyrin complex was grafted on B-doped p-Si photocathode for CO<sub>2</sub> reduction [127], which exhibited a significant improvement for CO evolution with faradaic efficiency of 80% after 6 h of reaction by effectively suppressing HER. As shown in Fig. 10(a), a molecular manganese catalyst (*fac*-[MnBr(4,4'-bis(phosphonicacid)-2,2'-bipyridine)(CO)<sub>3</sub>]) was integrated with the TiO<sub>2</sub> electrode to achieve a 67.5% faradaic efficiency during the PEC CO<sub>2</sub>-to-CO reduction [120]. Moreover, this work revealed the dynamic reconstruction of the Mn-Mn dimer active site on the TiO<sub>2</sub> electrode surface. Despite the metal complex-based cocatalysts have shown promising performance in PEC CO<sub>2</sub> conversion, the products mainly focus on two-electron reduction products like CO and HCOOH and higher-value hydrocarbon products (such as CH<sub>3</sub>OH and CH<sub>4</sub>) are rarely reported. Very recently, a trimethylammonio functionalized iron tetraphenylporphyrin complex was reported to produce eight-electron reduction product of CH<sub>4</sub> by integrating with Ir-based light sensitizer [128]. In which the CO was initially achieved and then converted into CH<sub>4</sub> with an impressive selectivity of ~82%. Therefore, the metal complex-based cocatalyst for PEC CO<sub>2</sub> reduction can be further developed to produce higher-value hydrocarbons with better efficiency.

#### 5.4.2 Metal-free cocatalysts

Besides the metal-based cocatalysts, the metal-free cocatalysts like organic molecules and polymers have also shown promising performances for PEC CO<sub>2</sub> reduction. An representative example is the dissolved tetraalkylammonium (NR<sup>4+</sup>, R = hydrocarbyl) salts on photocathodes, which can improve the activity and modulate the selectivity during PEC CO<sub>2</sub> reduction [129, 130]. This is because the NR<sup>4+</sup> species in electrolyte can adsorb on photocathodes and block the adsorption of H<sub>2</sub>O and then hinder the competing HER process. Moreover, the NR<sup>4+</sup> ions can capture electrons from photocathode and then promote the activation of CO<sub>2</sub> into -CO<sub>2</sub> by forming NR<sub>4</sub> radical intermediates.

The conductive polymers such as polypyrrole and polycation have been proven as effective cocatalysts for improving PEC CO<sub>2</sub> reduction, because the rich functional



**Fig. 10** (a) Schematic of molecular Mn cocatalyst on mesoporous TiO<sub>2</sub> for PEC CO<sub>2</sub> reduction. Reproduced with permission from Ref. [120]. (b) Schematic illustrations of PDDA/CdTe QDs or PMAEMA/CdTe QDs for PEC CO<sub>2</sub> reduction. Reprinted with permission from Ref. [121]. (c) A tandem PEC enzyme-cascade electrolyzer including the Rh complex and three dehydrogenases for PEC CO<sub>2</sub> conversion to CH<sub>3</sub>OH. Reproduced with permission from Ref. [122].

groups in polymers can promote the CO<sub>2</sub> reduction selectivity by binding with reaction molecules and modulating the adsorption of reaction intermediates [121, 131]. For instance, Isaacs *et al.* explored the behaviors of poly-diallyldimethylammonium (PDDA) and poly(2-trimethylammonio)ethyl methacrylate (PMAEMA) in PEC CO<sub>2</sub> reduction system by combining with CdTe quantum dots (QDs) via an electric field directed layer-by-layer assembly (EFDLA) method [Fig. 10(b)] [121]. The CO<sub>2</sub> reduction activity is related to the structure of the polycation and its assembly with QDs. Moreover, the distribution of reduction products is also associated with

the polycation type. Specifically, the PDDA/QDs assembly lead to the formation of CO, HCHO, and CH<sub>3</sub>OH but PMAEMA/QDs assembly trigger the conversion to HCHO.

### 5.4.3 Biological cocatalysts

Recently, many works have reported the developments on natural enzymes as promising cocatalysts for PEC CO<sub>2</sub> reduction [122, 132, 133]. The loading of biological cocatalysts onto semiconductors can provide highly active centers for PEC CO<sub>2</sub> reduction, leading to high performance and good selectivity for hydrocarbon products. For example, Park *et al.* constructed a tandem PEC electrolyzer by integrating enzyme-cascade (TPIEC), which can transfer photo-induced electrons to a multienzyme cascade for the conversion of CO<sub>2</sub> to methanol. In that reaction system, the nicotinamide cofactor (NADH) was consumed by dehydrogenases to produce NAD<sup>+</sup>, and then the NAD<sup>+</sup> was reduced to NADH. Under external bias, this TPIEC system exhibited an impressive CO<sub>2</sub>-to-methanol conversion rate of 1280 μmol·g<sup>-1</sup>·h<sup>-1</sup> [Fig. 10(c)] [122]. Armstrong and co-workers also developed a carbon monoxide dehydrogenase (CODH) from carboxydotherrus hydrogenoformans as a cocatalyst for a dye-sensitized p-NiO semiconductor and used it in PEC CO<sub>2</sub> reduction, which showed good selectively to reduce CO<sub>2</sub> into CO under visible-light [132, 133].

## 6 Conclusions and prospects

Recent progresses on the design of photoelectrode for PEC water splitting and CO<sub>2</sub> reduction are summarized and discussed in this review article. In the context of PEC water splitting, many research efforts have been focused on enabling more visible light utilization by the transition metal oxide semiconductors through nanostructuring and energy band engineering. However, the achievement of high photocurrent density often involves the use of noble metals. A few potential directions for future research could be:

(1) PEC water splitting cell under neutral or close to neutral conditions. Most existing PEC cells use acidic or basic solution, which pose potential hazards especially when leakage occurs.

(2) The use of noble metal in photoelectrode design can provide mechanistic understanding of the underlying photoelectrochemistry processes. However, in view of the end goal of deploying the PEC water splitting cell, cheap and earth abundant cocatalyst should be the focus.

(3) Although only a few common transition metal oxide semiconductors are reviewed in this article, new and emerging materials should be explored and investigated with the knowledge built upon these well-studied semiconductor materials.

In the context of PEC CO<sub>2</sub> reduction, despite a series of

exciting progresses have been achieved, our understanding of the PEC CO<sub>2</sub> reduction process remains limited. There are still many challenges:

(1) In the currently widely used cocatalysts for PEC CO<sub>2</sub> reduction system, most of the catalyst are based on the high-cost and scarce noble metals. Developing alternative cocatalysts that are based on earth-abundant and low-cost elements are imperative. To date, the Cu-based catalysts show promising effects in PEC CO<sub>2</sub> reduction, but they have low selectivity and usually produce a wide distribution of products. Therefore, improving the selectivity of Cu-based catalysts is of great significance.

(2) Improving PEC conversion efficiency of solar energy to CO<sub>2</sub> is necessary. High-efficiency of light utilization is important in increasing CO<sub>2</sub> reduction yield.

(3) The fundamental but complicated mechanism in PEC CO<sub>2</sub> reduction on a photocathode need to be further investigated, such as identifying the catalytic sites and reaction intermediates, which will better guide the development of highly effective catalysts.

**Contributions** Y. H. L. and B. Z. mainly worked on the solar water splitting part and solar CO<sub>2</sub> reduction part, respectively. All authors contributed to the writing of the manuscript.

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