

RESEARCH ARTICLE

β -PtO₂: Phononic, thermodynamic, and elastic properties derived from first-principles calculations

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β -PtO₂ is a useful transition metal dioxide, but its fundamental thermodynamic and elastic properties remain unexplored. Using first-principles calculations, we systematically studied the structure, phonon, thermodynamic and elastic properties of β -PtO₂. The lattice dynamics and structural stability of β -PtO₂ under pressure were studied using the phonon spectra and vibrational density of states. The vibrational frequencies of the optical modes of β -PtO₂ increase with elevating pressure; this result is comparable with the available experimental data. Then, the heat capacities and their pressure responses were determined based on the phonon calculations. The pressure dependence of the Debye temperature was studied, and the results were compared in two distinct aspects. The elastic moduli of β -PtO₂ were estimated through the Voigt–Reuss–Hill approximation. The bulk modulus of β -PtO₂ increases linearly with pressure, but the shear modulus is nearly independent of pressure. Our study revealed that the elastic stiffness coefficients C_{44} , C_{55} and C_{66} play a primary role in the slow variation of the shear modulus.

Keywords phonons, thermodynamic and elastic properties, β -PtO₂, first-principles calculations

1 Introduction

In recent years, transition metal oxides have been extensively studied owing to their unique electronic structures and thermodynamic and elastic properties [1–3]. Among these compounds, PtO₂ has received particular attention as a general-purpose catalyst with wide applications in optics, electrochemistry, and organic syntheses [4–6]. The rich body of experimental works on PtO₂, such as the synthesis of water-oxidizing catalysts [7] and the reaction behavior of PtO₂ in hydrosilylation [8], is mainly focused on practical applications. To date, the lattice dynamics and thermodynamic properties of PtO₂ are little known, although the electronic properties have been well reported [9–11]. Experimental studies have determined three crystalline phases of PtO₂ [12–14]: α -PtO₂, β -PtO₂, and β' -PtO₂, which have hexagonal CdI₂-, orthorhombic CaCl₂-, and tetragonal rutile-type structures, respectively. Unlike other Group VIII metal dioxides, all of which have a rutile structure [15], PtO₂ is the most stable in the beta phase (β -PtO₂) under ambient conditions. Based on the results of density of states, Wu and Weber concluded that this anomaly originates from strong hybridization of the Pt 5d and O 2p states at the Fermi level [9]. A compre-

hensive knowledge of the fundamental properties of PtO₂ is a prerequisite for elucidating the role and microscopic mechanism of PtO₂ in applications in fields such as optics and catalysis.

Motivated by the lack of basic properties in the existing literature, we systematically studied the phononic, thermodynamic, and elastic properties of β -PtO₂ through first-principles calculations. The specific heat, Debye temperature, and elastic modulus were determined for the first time, and the optical vibrational modes of β -PtO₂ are comprehensively investigated. Our calculations show that β -PtO₂ is stable over a wide pressure range. Increasing the applied pressure increases the vibrational frequencies of β -PtO₂ and decreases the heat capacity by hardening the atomic vibrations. The macroscopic effective elastic constants are deduced from the phononic and specific-heat calculations and compared with those derived from the single-crystal elastic stiffness. The calculated effective shear moduli of polycrystalline systems of β -PtO₂ are nearly pressure-independent owing to the slow variation of the elastic stiffness coefficients C_{44} , C_{55} , and C_{66} .

2 Computational methods

The structural optimization and total energy calculations were performed using the Vienna ab-initio Simulation

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Package (VASP) [16, 17], which is based on density functional theory (DFT). The exchange-correlation interactions of electrons were described using the local density approximation (LDA) [18, 19] and compared with the results of the generalized gradient approximation (GGA) using the Perdew–Burke–Erzerhoff (PBE) type functional [20]. Using a plane wave basis set the Kohn–Sham equations were solved, with the projected augmented wave (PAW) method [21] to describe the electron-ion core interactions. Energy cutoff of the plane-wave basis set is 600 eV. The Brillouin zone (BZ) was sampled on a $16 \times 16 \times 16$ Monkhorst–Pack k-mesh [22]. The vibrational and thermodynamic properties of β -PtO₂ were studied by combining the density functional perturbation theory (DFPT) as implemented in VASP with the direct method implemented in PHONOPY package [23]. Phonon calculations were performed in a $2 \times 2 \times 2$ supercell on a $6 \times 6 \times 6$ k-mesh. The lattice constants and phonons were calculated by both LDA and GGA for comparison. Additionally, the reliability of the standard DFT–GGA calculations was ensured via two methods: the first is the GGA + U method [24], and the second method involves hard pseudopotentials. The hard pseudopotentials were constructed in the pseudo-atomic configurations $5p^65d^96s^1$ and $2s^22p^4$ for Pt and O, respectively, with the plane-wave energy cutoff set to 950 eV. In the GGA + U calculation, the effective on-site Coulomb repulsion (U_{eff}) of the 5d orbitals of Pt was set to 7.5 eV [25]. The thermodynamic and elastic properties were calculated via only GGA.

3 Results and discussion

3.1 Structural properties

Figure 1 shows the crystal structure and simulated X-ray diffraction pattern of β -PtO₂. The results indicate that β -PtO₂ has an orthorhombic CaCl₂-type crystal structure, which is similar to the rutile structure but with a slight

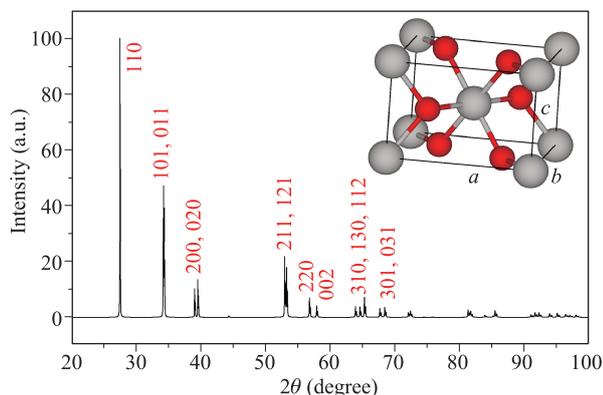


Fig. 1 Crystal structure and simulated X-ray diffraction (Cu $K\alpha 1$ radiation, wavelength ~ 1.541 Å) pattern of β -PtO₂. The large gray and small red balls represent Pt and O atoms, respectively.

distortion in the directions of the cell axes a and b . Two Pt atoms and six O atoms are located in the $2a$ and $4g$ Wyckoff positions, respectively. The Pt atoms are surrounded by four and two oxygen atoms at 2.043 Å and 2.036 Å, respectively. The strongest signal in the X-ray diffraction data was found at $2\theta = 27.49^\circ$, with a d spacing of 3.24 Å. The corresponding diffraction plane is (110). The second strongest diffraction appears in the (101) and (011) planes in the 34.19° – 34.41° and 2.61–2.62 Å ranges of incident angles and d -spacing, respectively.

We calculated the total energy E by changing the lattice volume V to get a series of E – V data, and then the data are least squares fitted to the Murnaghan equation of state [26] to obtain the elastic properties:

$$E_{\text{tot}}(V) = \frac{B_0 V}{B'_0} \left[\frac{(V_0/V)^{B'_0}}{B'_0 - 1} + 1 \right] + E_0 - \frac{B_0 V_0}{B'_0 - 1}, \quad (1)$$

where E_0 and V_0 are the total energy and volume at equilibrium, while B_0 and B'_0 are the bulk modulus and its pressure derivative, respectively. The results are shown in Fig. 2, and the calculated data are listed in Table 1. The calculated lattice constants are in good agreement with the experimental and other theoretical data [15, 27]. The results calculated through the GGA + U computation differ from that calculated through the standard method by less than 5%, indicating that the choice of the Hubbard term U does not significantly affect the main properties of β -PtO₂. The deduced values of B_0 and B'_0 are 209 GPa and 4.06, respectively. To our knowledge, the first set of data on the bulk modulus of β -PtO₂ is reported in this paper. The pressure was determined using the Murnaghan equation of state:

$$P = - \left(\frac{\partial E}{\partial V} \right) = (B_0/B'_0) \left[(V_0/V)^{B'_0} - 1 \right], \quad (2)$$

Accordingly, we can obtain the volume and energy at any pressure point within a certain range, e.g., 0–30 GPa in our study.

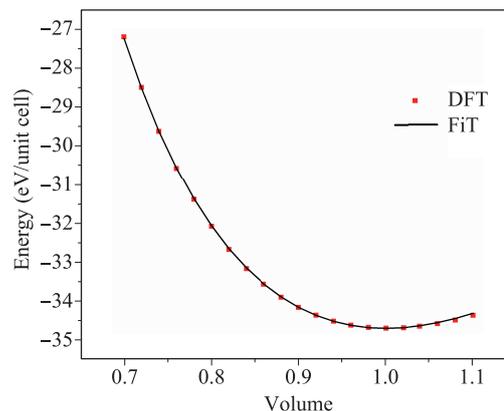


Fig. 2 Total energy of β -PtO₂ as a function of volume (normalized to the equilibrium volume at $P = 0$). The solid curve is the data fitted according to the Murnaghan equation of state.

Table 1 Calculated lattice constants, volumes, bulk moduli, and pressure derivatives of the bulk modulus.

	a (Å)	b (Å)	c (Å)	V_0 (Å ³)	B_0 (GPa)	B'_0
GGA	4.613	4.557	3.185	66.95	209	4.06
LDA	4.492	4.442	3.142	62.69	265	3.90
GGA + U	4.557	4.569	3.136	65.30	220	3.81
Experiment [15]	4.484	4.539	3.136	63.83		
Other works [27]	4.61	4.55	3.19	66.91		

3.2 Phonon properties

Figure 3 illustrates the optical vibrational modes of β -PtO₂ calculated via the DFPT method at the gamma point. For simplicity, we refer to these modes as ω_1 – ω_{15} , where the frequency ranges from high (ω_1) to low (ω_{15}). Note that ω_2 , ω_4 , ω_7 , ω_8 , ω_{10} , and ω_{12} represent the vibrations of O atoms only, which are Raman-active modes, and consistent with the experimentally measured Raman spectra [28]. As the other modes of β -PtO₂ lack Raman activity, they must be measured via other detection methods, such as infrared spectroscopy. Table 2 compares the

calculated frequencies of the optical modes of β -PtO₂ and the experimental data. In general, the frequencies were higher in the LDA calculation than in the GGA calculation; therefore, the LDA values better matched the experimental data in the high-frequency region. However, the frequencies calculated through GGA were closer to the experimental values in the low-frequency region. The phonon modes of β -PtO₂ can be roughly divided into three categories. The high-frequency modes of β -PtO₂, ω_1 – ω_8 ($500 < \omega < 800$ cm⁻¹), are attributed to the coupled modes that mainly involve the stretching of the Pt–O bonds. The mid-frequency modes of β -PtO₂, ω_9 – ω_{12} ($250 < \omega < 450$ cm⁻¹), correspond to the O–Pt–O bending vibrations in the PtO₆ octahedron. Finally, the low-frequency modes of β -PtO₂, ω_{13} – ω_{15} ($\omega < 250$ cm⁻¹), originate from the interactions between different Pt polyhedrons.

To study the atomic-scale dynamical properties, we calculated the phonon dispersion curve and vibrational density of states (VDOS) of β -PtO₂ along some high-symmetry points through both LDA and GGA for pressures up to 30 GPa. The results are displayed in Fig. 4. The phonon spectra and VDOS indicate that β -PtO₂ is dynamically stable over the pressure range 0–30 GPa. In-

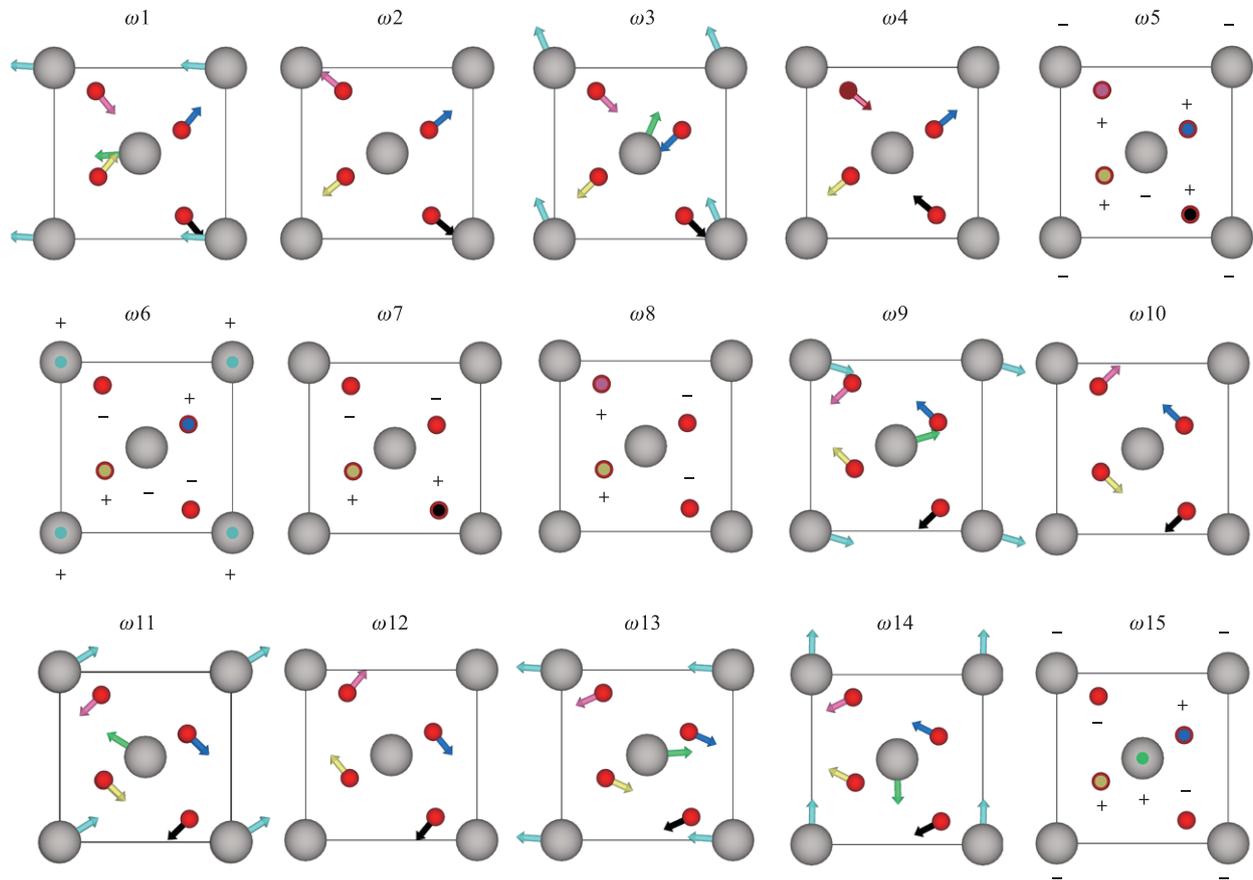


Fig. 3 Top views of eigenvectors of the optical vibrational modes of β -PtO₂. The gray and red balls represent the Pt and O atoms, respectively. The symbols + and – indicate that the vibration directions are upward and downward, respectively, along the c axis.

Table 2 Calculated frequencies of the optical vibrational modes of β -PtO₂ and (for comparison) the available experimental data.

Mode	Frequency (cm ⁻¹)			Symmetry (D _{2h})
	LDA	GGA	Expt. ¹⁾	
ω_1	728	665		B_{1g}
ω_2	716	651	(~ 790)	
ω_3	700	635		
ω_4	697	627	743	A_g
ω_5	683	626		B_{3g}
ω_6	677	624		
ω_7	582	532	616	
ω_8	561	512	606	B_{2g}
ω_9	402	383		B_{1g}
ω_{10}	391	371	340	
ω_{11}	378	358		
ω_{12}	279	254	205	A_g
ω_{13}	223	208		
ω_{14}	222	202		
ω_{15}	183	167		

¹⁾Ref. [28].

creasing the pressure shifted all optical phonon modes of β -PtO₂ upward (toward higher frequencies). The GGA better described the previous experimentally measured vibrational frequencies in the acoustic and low-frequency optical phonon branches, and the LDA better described the high-frequency optical branches. In addition, the gaps between some dispersion bands in the high-frequency re-

gion were enlarged under pressure, indicating the different pressure responses of the frequency modes of β -PtO₂.

3.3 Thermodynamic properties

The heat capacity (specific heat in the case of unit volume, mass, or number of particles) is a basic thermodynamic quantity that describes the extent to which a substance absorbs and releases heat. From a microscopic viewpoint, the heat capacity of a substance manifests owing to the thermal motions of atoms and electrons. In semiconductors and insulators, the main contribution to heat capacity is atomic vibrations, i.e., phonons. Based on first-principles phonon calculations and statistical mechanics, the constant-volume heat capacity is given as follows:

$$C_V = \left(\frac{\partial E}{\partial T} \right)_V = \sum_{qv} k_B \left[\frac{\hbar\omega(qv)}{k_B T} \right]^2 \frac{\exp(\hbar\omega(qv)/(k_B T))}{[\exp(\hbar\omega(qv)/(k_B T)) - 1]^2}, \quad (3)$$

where \hbar is the reduced Planck constant, k_B is the Boltzmann constant, T , ω , and q denote the temperature, frequency and wave vector, respectively, and the index ν distinguishes the various phonon branches. C_V is well approximated using the Debye model, which describes the change in heat capacity with temperature. The general form of the Debye model is as follows [29]

$$C_V = 9N_A k_B \left(\frac{T}{\Theta_D} \right)^3 \int_0^{\Theta_D/T} \frac{x^4 e^x}{(e^x - 1)^2} dx, \quad (4)$$

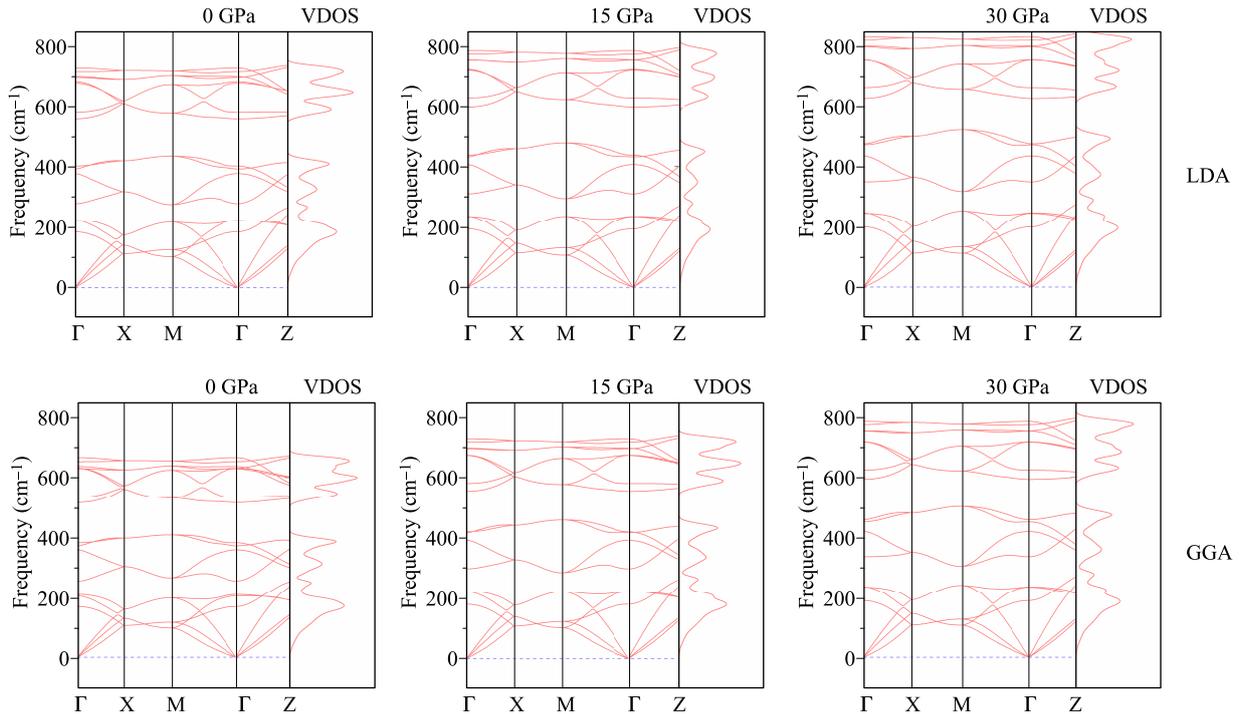


Fig. 4 Calculated phonon dispersions along some high-symmetry lines of the BZ, and the VDOS of β -PtO₂ under different pressures.

where N_A is the Avogadro constant, Θ_D is the Debye temperature, and the other parameters have their usual meanings. In the low-temperature limit ($T \rightarrow 0$ when $\Theta_D/T \rightarrow \infty$), the Debye model reduces C_V to $\sim \frac{12Nk_B\pi^4}{5} \left(\frac{T}{\Theta_D}\right)^3$ [29, 30], which is proportional to T^3 ; this is called the Debye T^3 law. To verify the applicable range of this law, we compared the specific heat obtained through DFT calculations (which are particularly accurate at room temperature and below) with the curve fitted to the Debye model (see Fig. 5). By fitting the specific-heat curve to Eq. (3), we estimated the Debye temperature of β -PtO₂ as ~ 512 K.

At temperatures below 20 K, the two curves are almost identical. Above 20 K, the Debye model deviates significantly from the specific heat predicted by first-principles calculations. Therefore, the Debye T^3 law applies when $T \leq \frac{1}{30}\Theta_D$. In fact, the value of the integral $\int_0^{30} \frac{x^4 e^x}{(e^x - 1)^2} dx$ differs by only $\sim 1.5 \times 10^{-8}$ from the exact value ($\int_0^\infty \frac{x^4 e^x}{(e^x - 1)^2} dx = \frac{4\pi^4}{15}$ at the low temperature limit), justifying the aforementioned temperatures at which the Debye model can be applied ($T \leq \frac{1}{30}\Theta_D$). On the other hand, in the high temperature region, the calculated specific heat approaches the classical thermodynamic limit of $3R$ [31].

Figure 6(a) plots the temperature dependence of the specific heat and its variations with pressure. The specific heat gradually increases and decreases with increasing temperature and pressure, respectively; this occurs because the compression of crystal volume enhances the lattice vibrations and increases the vibrational frequencies, thus reducing the heat absorption in the same vibrational mode. Figure 6(b) shows the pressure dependence of some typical vibrational frequencies. The maximum frequency of the optical modes at the gamma point, Debye frequency, and maximum frequency of the acoustic branches monotonically increase with pressure because the interatomic force constants increase with pressure. The Debye

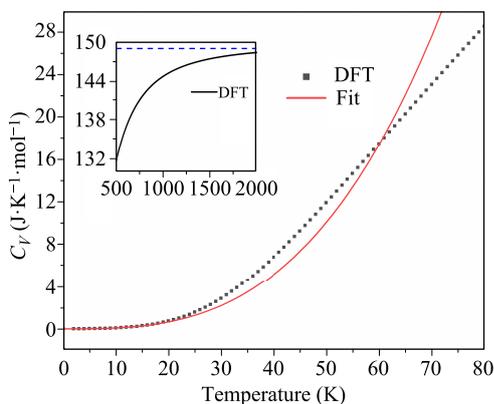


Fig. 5 Calculated specific heat of β -PtO₂ at low temperatures, and (for comparison) the heat capacity fitted using the Debye model. The inset shows the high-temperature behavior of the specific heat.

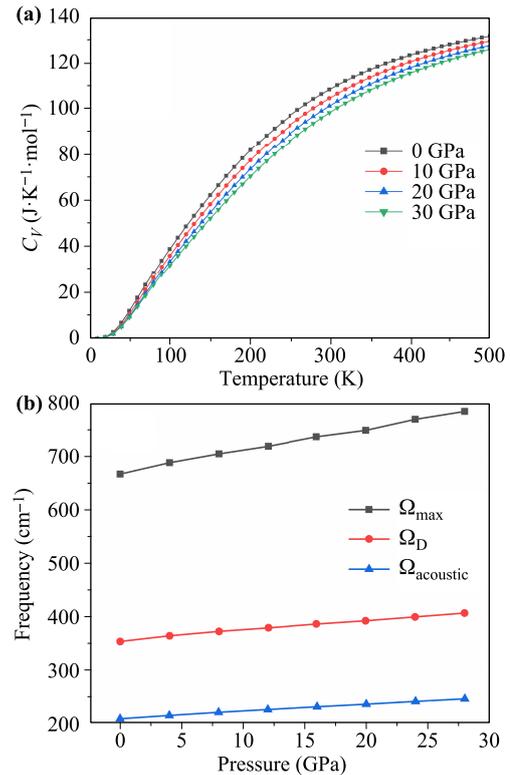


Fig. 6 (a) Calculated heat capacity of β -PtO₂ under different pressures. (b) Maximum phonon frequency of the gamma point, Debye frequency, and maximum acoustic frequency as functions of pressure.

frequency is located between the maximum frequency of the gamma point (i.e., the maximum frequency of the optical mode of β -PtO₂ in the long wavelength limit) and that of the acoustic branch (deduced from the VDOS). The numerical values of these three frequencies are quite different, indicating that these two frequencies cannot simply replace the Debye frequency in the Debye model. Moreover, the Debye frequency is almost proportional to the acoustic branch frequency, which is consistent with the fact that the acoustic phonons are the main contributors to the low-temperature specific heat.

3.4 Elastic properties

The mechanical and thermodynamic behavior of materials can be characterized by analyzing their elastic properties on the atomic scale. The elastic constants, which reflect the response of the crystal to external strain, are important for understanding the structural stability and strength of a substance. β -PtO₂ is structurally orthorhombic with nine independent elastic stiffness coefficients: C_{11} , C_{22} , C_{33} , C_{44} , C_{55} , C_{66} , C_{12} , C_{13} , and C_{23} [32]. The effective bulk modulus B and shear modulus G of β -PtO₂ polycrystals can be obtained by combining these coefficients with the Voigt–Reuss–Hill approximation [33–35]:

$$B_V = (1/9)[C_{11} + C_{22} + C_{33} + 2(C_{12} + C_{13} + C_{23})],$$

$$\begin{aligned}
 G_V &= (1/15)[C_{11} + C_{12} + C_{13} + 3(C_{44} + C_{55} + C_{66}) \\
 &\quad - (C_{12} + C_{13} + C_{23})], \\
 B_R &= \Delta[C_{11}(C_{22} + C_{33} - 2C_{23}) + C_{22}(C_{33} - 2C_{13}) \\
 &\quad - 2C_{33}C_{12} + C_{12}(2C_{23} - C_{12}) \\
 &\quad + C_{13}(2C_{12} - C_{13}) + C_{23}(2C_{13} - C_{23})]^{-1}, \\
 G_R &= 15\{4[C_{11}(C_{22} + C_{33} + C_{23}) + C_{22}(C_{33} + C_{13}) \\
 &\quad + C_{33}C_{12} - C_{12}(C_{23} + C_{12}) - C_{13}(C_{12} + C_{13}) \\
 &\quad - C_{23}(C_{13} + C_{23})]/\Delta \\
 &\quad + 3[(1/C_{44}) + (1/C_{55}) + (1/C_{66})]\}^{-1}, \\
 \Delta &= C_{13}(C_{12}C_{23} - C_{13}C_{22}) + C_{23}(C_{12}C_{13} - C_{23}C_{11}) \\
 &\quad + C_{33}(C_{11}C_{22} - C_{12}^2), \\
 B &= 1/2(B_R + B_V), \quad G = 1/2(G_R + G_V), \quad (5)
 \end{aligned}$$

where the subscripts V and R denote the elastic moduli obtained via the methods of Voigt and Reuss, respectively. From B and G , the Young's modulus E and Poisson's ratio ν are obtained as follows [36]:

$$E = \frac{9BG}{3B + G}, \quad \nu = \frac{3B - 2G}{6B + 2G}. \quad (6)$$

To study these characteristics systematically, we calculated the elastic constants and related moduli parameters of β -PtO₂ at 0 K, varying the pressure from 0 to 30 GPa. The results of these calculations are listed in Table 3. The elastic stiffness coefficients increase with increasing pressure; the sole exception is C_{66} , which decreases with increasing pressure. Further, the bulk modulus B , Young's modulus E , and Poisson's ratio ν increase with increasing pressure, while the shear modulus G only fluctuates slightly with pressure, which exhibits no obvious trend.

The Debye temperature, an important thermodynamic parameter, is closely related to many basic properties of solids, such as the specific heat, thermal conductivity, and melting temperature. As the vibration excitation at low temperatures is mainly due to the acoustic modes of β -PtO₂, the Debye temperature of solids can be calculated by averaging the sound velocity v_m [37] as follows:

$$\Theta_e = \frac{h}{k_B} \left[\frac{3n}{4\pi} \left(\frac{N_A \rho}{M} \right) \right]^{1/3} v_m, \quad (7)$$

Table 3 Calculated elastic stiffness coefficients C_{ij} (GPa), bulk modulus B (GPa), shear modulus G (GPa), Young's modulus E (GPa), and Poisson's ratio ν of β -PtO₂ at different pressures.

pressure (GPa)	C_{11}	C_{22}	C_{33}	C_{44}	C_{55}	C_{66}	C_{12}	C_{13}	C_{23}	B	G	E	ν
0	272	242	413	171	101	64	185	143	140	204.5	85.6	225.0	0.316
4	286	254	431	180	102	63	201	155	150	217.7	86.3	228.6	0.325
8	304	266	456	184	104	60	218	172	159	232.8	86.4	230.7	0.335
12	328	280	469	190	105	57	230	187	170	246.5	88.4	236.9	0.340
16	344	293	486	196	106	53	246	202	181	260.7	88.0	237.3	0.348
20	365	306	502	199	107	50	260	218	190	274.2	88.6	240.0	0.354
24	375	319	525	204	106	45	278	232	202	288.9	86.3	235.5	0.364
30	415	344	541	208	109	44	292	254	216	308.6	91.0	248.6	0.366

where h is the Planck constant, n is the number of atoms per unit cell, ρ is the density of the material, M is the relative atomic mass, and the other parameters have the usual meanings as above. In the Debye model, the average sound velocity of a material is given as follows:

$$v_m = \left[\frac{1}{3} \left(\frac{2}{v_l^3} + \frac{1}{v_t^3} \right) \right]^{-1/3}, \quad (8)$$

where v_l and v_t are the longitudinal and transverse sound velocities, respectively; these can be derived from Navier's equation [29]:

$$v_l = \sqrt{\frac{3B + 4G}{3\rho}}, \quad v_t = \sqrt{\frac{G}{\rho}}. \quad (9)$$

Further, we calculated the phonon-determined Debye temperature (Θ_ω) by fitting the low-temperature specific heat to the Debye T^3 law and compared it with the acoustic Debye temperature (Θ_e) obtained from Eq. (7). The results of this comparison are plotted in Fig. 7 as functions of pressure. Unexpectedly, the two plots were quite different. The phonon-determined Debye temperature increased almost linearly with increasing pressure, whereas the acoustic Debye temperature was relatively uniform and insensitive to pressure.

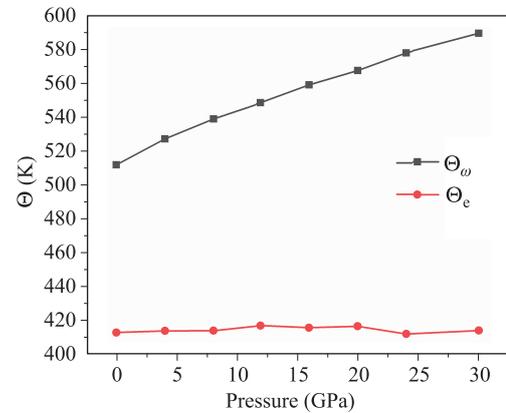


Fig. 7 Pressure dependences of the phonon-determined and acoustic Debye temperatures (Θ_ω and Θ_e , respectively).

To find the cause of this difference, we plot the pressure responses of the bulk modulus B , shear modulus G , and sound velocity in Figure 8. The macroscopic effective bulk modulus, B_{fit} , was obtained by fitting the $E-V$ data to the Murnaghan equation of state. The macroscopic effective shear modulus, G_{fit} , and fitted sound velocities were derived from B_{fit} and the phonon-determined Debye temperatures through Eqs. (7)–(9). Both the Voigt and Reuss approximations yield good results for the bulk modulus, but the shear moduli computed using both approximations largely deviated from G_{fit} . This discrepancy originates the fact that shear modulus is a macroscopic quantity, which is only applicable to isotropic systems in principle; however, the system under consideration is an orthorhombic one. Consequently, the Debye temperatures calculated by fitting the low-temperature specific heat of first-principles are suitable for single-crystal systems, and the values estimated using Eq. (7) are applicable to polycrystalline systems. Similarly, the effective elastic moduli (B_{fit} , G_{fit}) deduced from the low-temperature specific heat are only applicable to single-crystals of β -PtO₂, while the ones calculated using the Voigt–Reuss–Hill approximation are suitable for the polycrystalline phases of β -PtO₂.

As shown in Fig. 8(b), the effective shear moduli (G_V

and G_R) vary much more slowly with pressure than the bulk moduli (B_V , B_R). The pressure derivatives of the average bulk and shear moduli were 3.54 and 0.07, respectively. Additionally, the pressure variations in the longitudinal and transverse sound velocities were $\sim 17 \text{ m}\cdot\text{s}^{-1}\cdot\text{GPa}^{-1}$ and $-3.4 \text{ m}\cdot\text{s}^{-1}\cdot\text{GPa}^{-1}$, respectively. The nearly constant shear moduli imply that the transverse sound velocity is nearly invariant within the pressure range 0–30 GPa [as shown in Fig. 8(d)]. Furthermore, we considered the effectiveness of the different pseudopotentials. The pressure variation of the elastic modulus obtained using the hard pseudopotential was consistent with that of the standard DFT–GGA calculations (the two results deviated within 5%). Therefore, standard GGA potentials are still valid for describing the β -PtO₂ system under high pressure (~ 30 GPa).

To explore the fundamental cause of the slowly changing shear modulus, we plotted the elastic stiffness coefficients as functions of pressure (Fig. 9). We observe that C_{66} decreases with increasing pressure and that C_{44} , C_{55} , and C_{66} change much more slowly than the other stiffness coefficients. Combined with the expression of the shear modulus [Eq. (5)], we infer that the slow variation of the shear modulus is mainly owing to the three stiffness coefficients, C_{44} , C_{55} , and C_{66} , which describe the weak shear motions

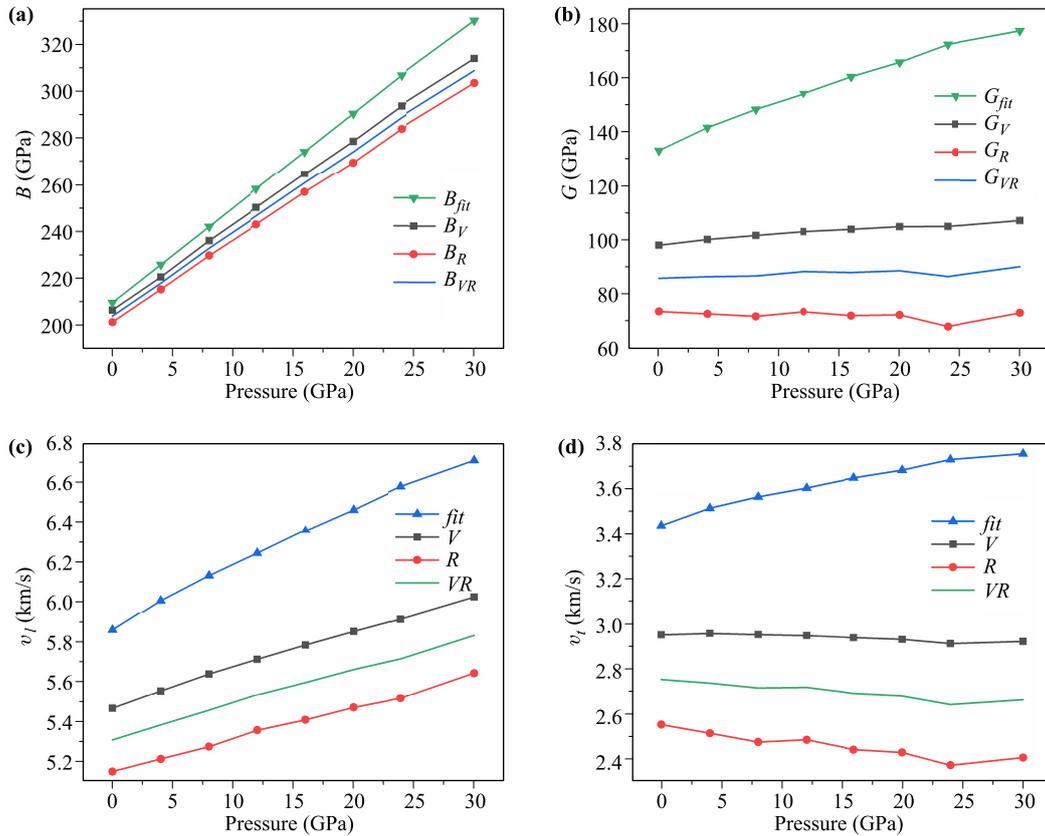


Fig. 8 (a) Bulk modulus, (b) shear modulus, (c) longitudinal sound velocity and (d) transverse sound velocity at different pressures. The subscript *fit* represents the moduli obtained by fitting Eq. (1) and Eqs. (7)–(9) using the data of specific heat, and the subscripts *V*, *R* and *VR* represent Voigt and Reuss approximation and their arithmetic average [Eq. (5)], respectively.

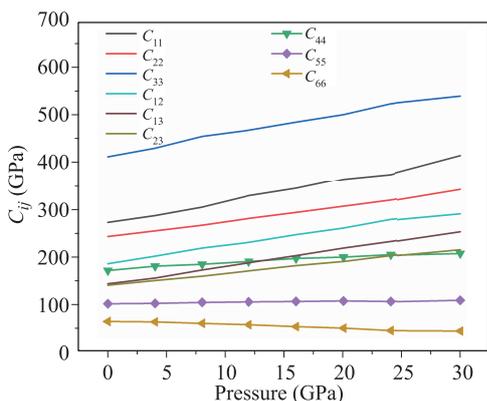


Fig. 9 Pressure dependence of the elastic stiffness coefficient C_{ij} .

of the xy , xz , and yz planes of the orthorhombic β -PtO₂. To our knowledge, the anomalies of the elastic coefficients C_{44} , C_{55} , and C_{66} in transition metal dioxide (TMO₂) systems have not been previously reported; however, such anomalies in cubic lattice materials (e.g., ZnAl₂O₄ and Fe₃O₄) have been reported. In these systems, the elastic coefficient C_{44} changes slowly or decreases with pressure [38, 39]. Such anomalous behavior occurs because of the relationship between the shear modulus and sliding moments at the grain boundaries of polycrystalline materials, which are less sensitive to volume compression of the composite crystallites. Therefore, we expect that the anomaly of shear modulus would also exist in the other systems such as tetragonal and hexagonal polycrystalline materials.

4 Conclusions

Based on the DFT calculations, we have systematically studied the structure and phononic, thermodynamic, and elastic properties of β -PtO₂ under different pressures. The calculated lattice parameters are in good agreement with those reported in previous works. An extensive investigation is presented on the characteristics of vibrational modes of β -PtO₂, which is found to be stable over a wide range of pressures. The phonon modes of β -PtO₂ in the high-frequency region were discretized and enhanced under pressure. The low-temperature specific heat was confirmed to follow the Debye T^3 law. The Debye temperatures were deduced from the specific-heat data (single-crystal) and macroscopic effective elastic constants. The two sets of results were compared in the pressure range 0–30 GPa. As evidenced by the notable difference between the calculated Debye temperatures, the results of anisotropic single-crystal systems are not applicable in isotropic polycrystalline systems. The pressure dependences of the effective elastic moduli were calculated and compared through the Voigt–Reuss–Hill approximation. The bulk and shear moduli responded differently to in-

creasing pressure: the bulk modulus increased uniformly whereas the shear modulus varied slowly. The slow variation of the shear modulus causes the transverse sound velocity to be nearly independent of pressure variations. We expect that these results can be further tested in future experimental studies.

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